

# Sulfur-Tolerant Anodes for SOFCs

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# Outline

- **Technical Issues Addressed**
- **Objectives & Approach**
- **Recent Progress (Since Oct 2004)**
  - The H<sub>2</sub>S Poisoning Effect
  - Thermodynamic Analysis
  - QM Calculations
  - Exploration of New Sulfur-Tolerant Anode Materials
  - Mechanisms of Sulfur-Anode Interactions
- **Activities for the next 6-12 Months**

# Critical Issues

- **What are the impacts of sulfur poisoning?**
- **How to study the interactions between sulfur and anodes?**
- **What is the mechanism of sulfur poisoning?**
- **What is required to achieve the sulfur tolerance needed for the SECA program?**
- **How to design new materials with required sulfur tolerance?**

# Objectives

- **To characterize the effect of sulfur-poisoning on fuel cell performance under various operating conditions**
- **To investigate the detailed mechanisms of sulfur-poisoning**
- **To develop strategies for achieving both sulfur-tolerance and high performance**
- **To explore new sulfur-tolerant materials to meet SECA Program objectives**

# Technical Approach

- **Phenomenological Characterization of Sulfur Poisoning Effect**
  - Impedance spectroscopy (I.S.)
  - Cell performance and anode over-potential
- **Understanding Sulfur Poisoning Mechanism**
  - *ex-situ* examination of the anode via XRD, Raman, etc
  - *in-situ* Raman spectroscopy coupled with I.S.
  - Thermodynamic/kinetic analysis
  - MD and QM calculations
- **Design of New Anode Materials/New Structure**
  - For modification/decoration of Ni-YSZ Surface
  - For replacement of Ni-YSZ anode

# Recent Progress: Since Oct 2004

- **The H<sub>2</sub>S Poisoning Effect**
- Thermodynamic Analysis
- QM Calculations
- Exploration of New Sulfur-Tolerant Anode Materials

# H<sub>2</sub>S Poisoning of Ni-YSZ

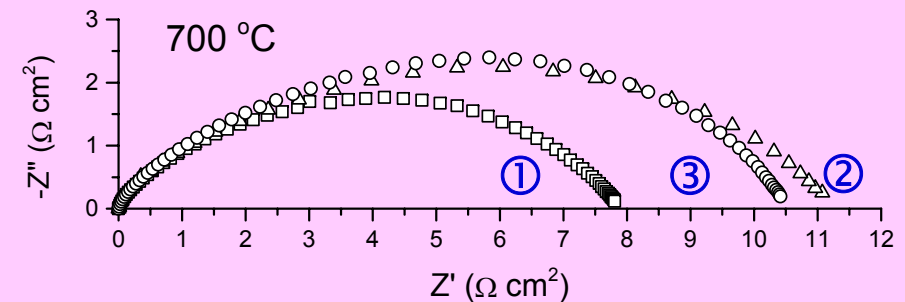
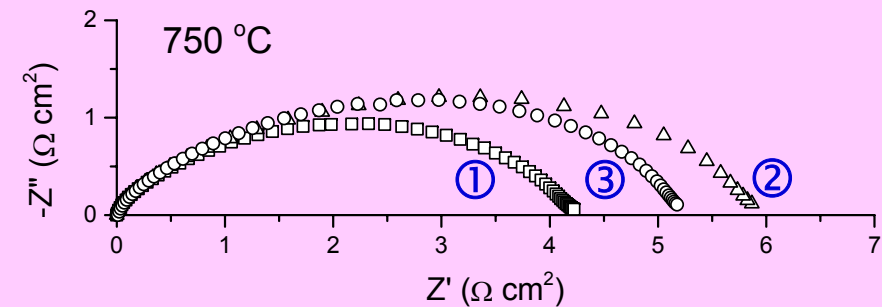
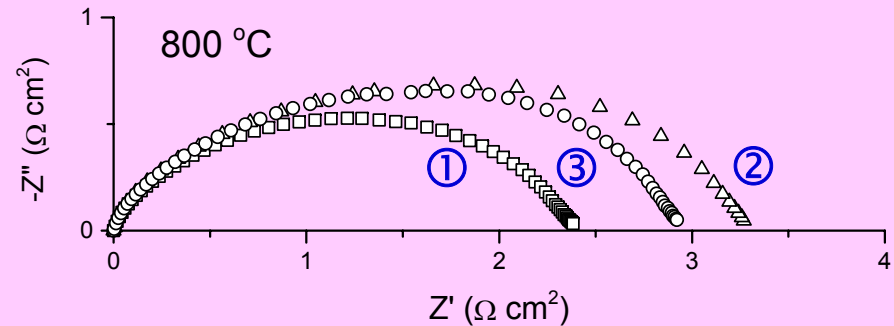
## Impedance Spectra

of a symmetrical cell

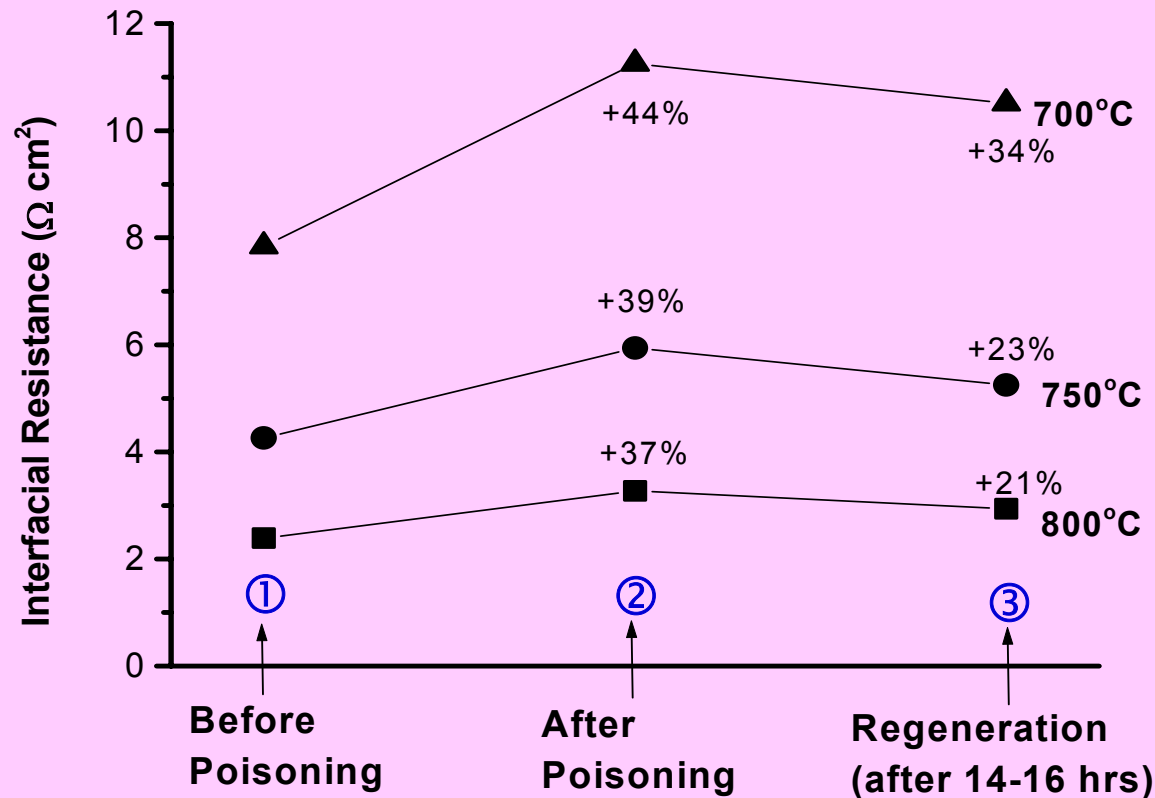
Ni-YSZ/YSZ/Ni-YSZ

immersed in

- ① Humidified H<sub>2</sub> for 3 days
- ② 50 ppm H<sub>2</sub>S (50v%H<sub>2</sub> & 50% N<sub>2</sub>) for 3 hours
- ③ Humidified H<sub>2</sub> (Regeneration) for 14-16 hours



# Summary: Effect of H<sub>2</sub>S on Rp



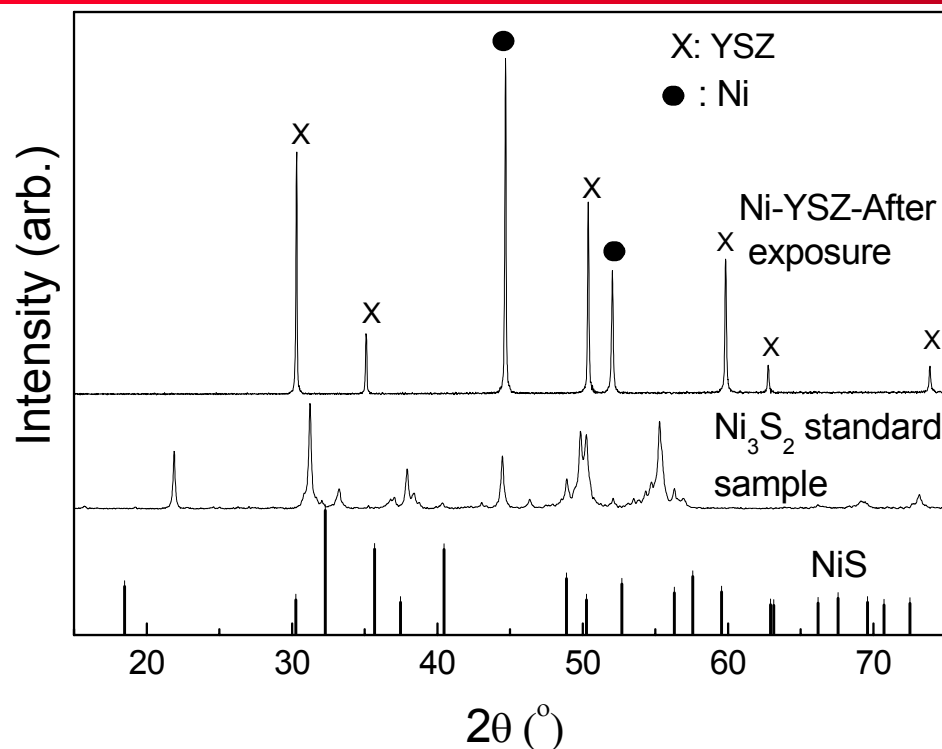
## At lower temperature

- Poisoning occurs faster and the effect is stronger
- Recovery is slower and more difficult

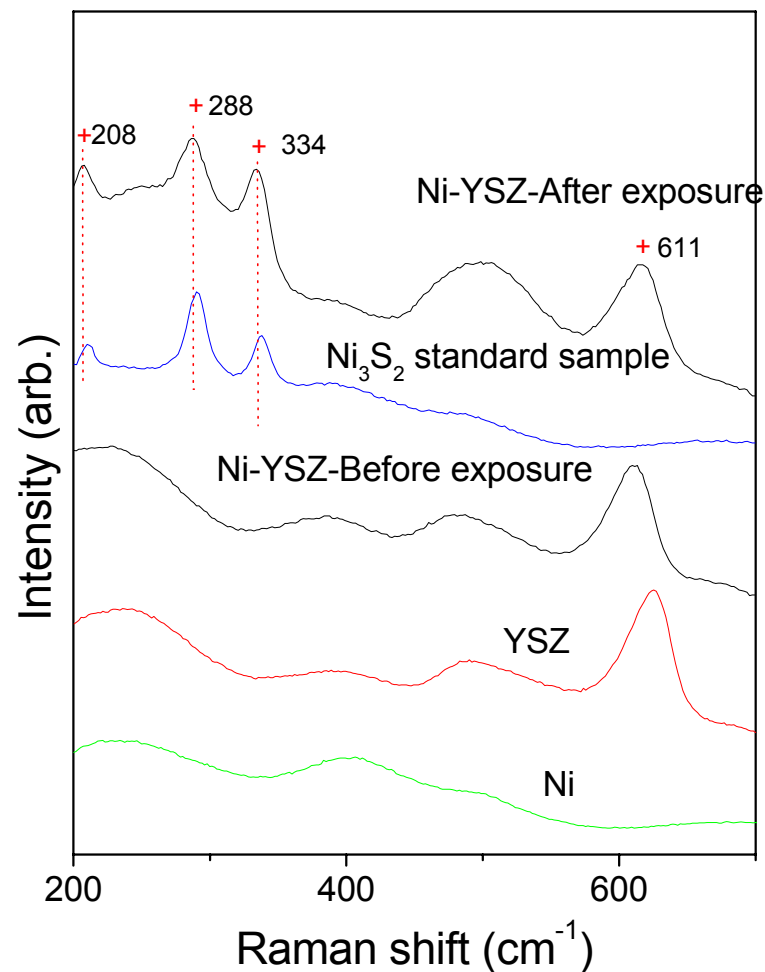
Note: The percentage adjacent to each data point represents the increase in Rp after poisoning or regeneration compared to the initial value before exposure to H<sub>2</sub>S.



# XRD and Raman Study of S-Ni Interactions



Ni-YSZ cermet was exposed to humidified hydrogen containing 100 ppm H<sub>2</sub>S at 727°C for 5 days.



**Bulk sulfides are not detected by XRD; yet S-Ni vibrations are identified by Raman**

# Implications

- **The surface sulfur is most likely responsible to the observed degradation in performance.**
- **While XRD is insensitive to sulfur poisoning, Raman spectroscopy could be used for probing and mapping of NiS<sub>x</sub> under in-situ conditions and hence for elucidating the sulfur-poisoning mechanism.**

# Question

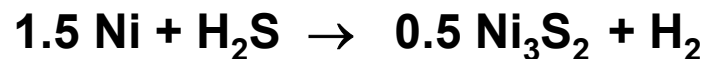
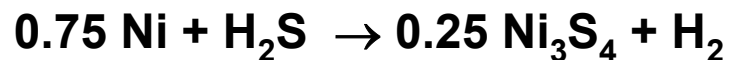
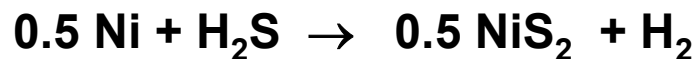
**What are the conditions under which nickel sulfides are expected?**

# Recent Progress: Since Oct 2004

- The H<sub>2</sub>S Poisoning Effect
- **Thermodynamic Analysis**
- QM Calculations
- Exploration of New Sulfur-Tolerant Anode Materials

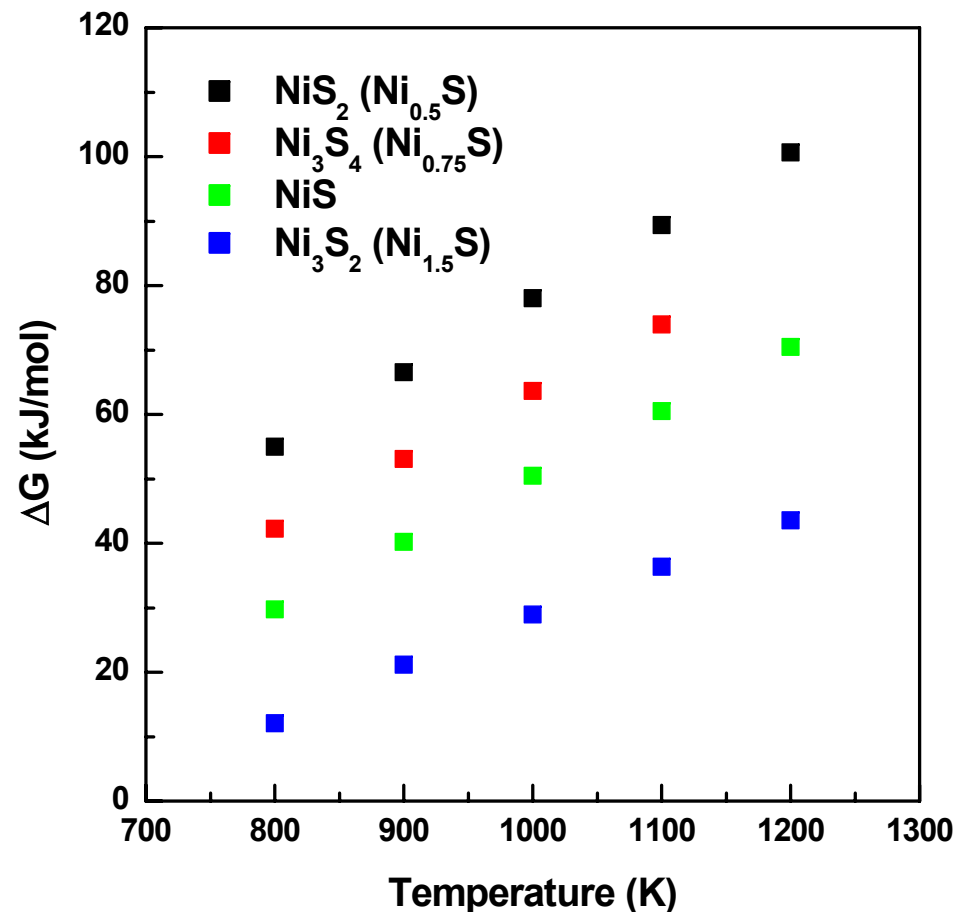
# Thermodynamic Stability of Ni in 50 ppm H<sub>2</sub>S

- Possible reactions between H<sub>2</sub>S and Ni (**bulk-phase**):



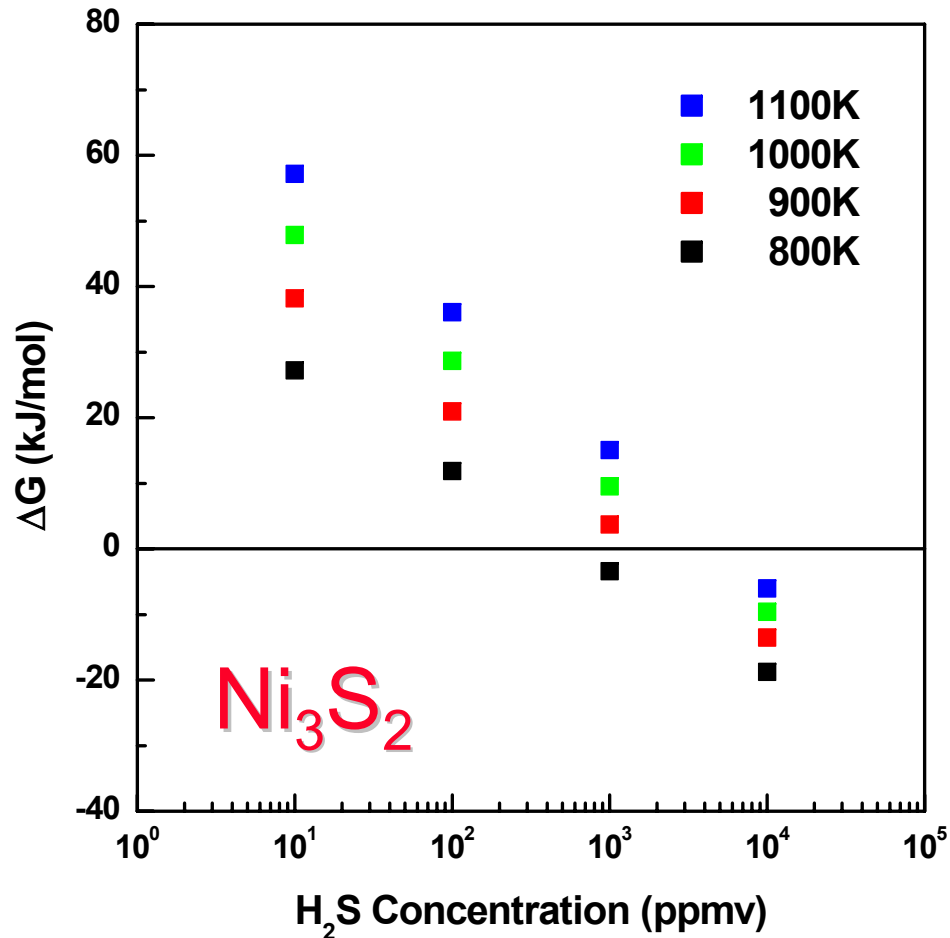
**Thermodynamics** predicts that (**bulk-phase**) Nickel is stable (or sulfide **Ni<sub>x</sub>S<sub>y</sub>** is unstable) in **50 ppm H<sub>2</sub>S** at elevated temperatures

The stability of **Ni<sub>x</sub>S<sub>y</sub>** increases with Ni to S ratio; Ni<sub>3</sub>S<sub>2</sub> detected



Reaction Gibbs free energy change for several possible reactions between **Ni** and **50 ppmv H<sub>2</sub>S** in 50%**H<sub>2</sub>**/1.5%**H<sub>2</sub>O**/48.5%**N<sub>2</sub>** at elevated temperatures

# Effect of H<sub>2</sub>S Concentration



Dependence on H<sub>2</sub>S concentration of the reaction Gibbs free energy for the formation of Ni<sub>3</sub>S<sub>2</sub> from the reaction between Ni and H<sub>2</sub>S in 97%H<sub>2</sub>/3%H<sub>2</sub>O

- The stability of (bulk-phase) Ni sulfide increases with H<sub>2</sub>S concentration but decreases with temperature
- Ni sulfide (bulk-phase) will form only at relatively high concentration of H<sub>2</sub>S at low temperatures  
e.g., Ni<sub>3</sub>S<sub>2</sub> is stable only when [H<sub>2</sub>S] > 10<sup>3</sup> ppmv at T < 800K

# Conclusion

- **Thermodynamics predicts that Ni-YSZ are stable in 50 ppm H<sub>2</sub>S at >700C.**
- **The thermodynamic analysis does not seem to be helpful in understanding what is happened to Ni exposed to 50 ppmv H<sub>2</sub>S.**

# Speculation/Hypothesis

- It was suspected that the **adsorption energy** for sulfur on Ni is significantly **higher (more negative) than the bonding energy** in a  $\text{Ni}_x\text{S}_y$  crystal\*.

- It appears that **sulfur adsorption** on Ni surface



is energetically favorable in low concentration of  $\text{H}_2\text{S}$  even when the formation of sulfides is unfavorable.

- \* C. H. Bartholomew, P. K. Agrawal, J. R. Katzer, "Sulfur Poisoning of Metals," *Advances in Catalysis*, Vol. 31 (1982), p 135.

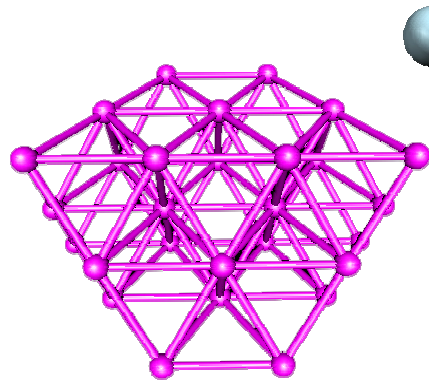
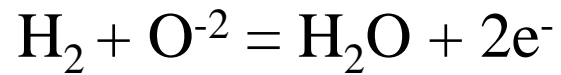


# Recent Progress: Since Oct 2004

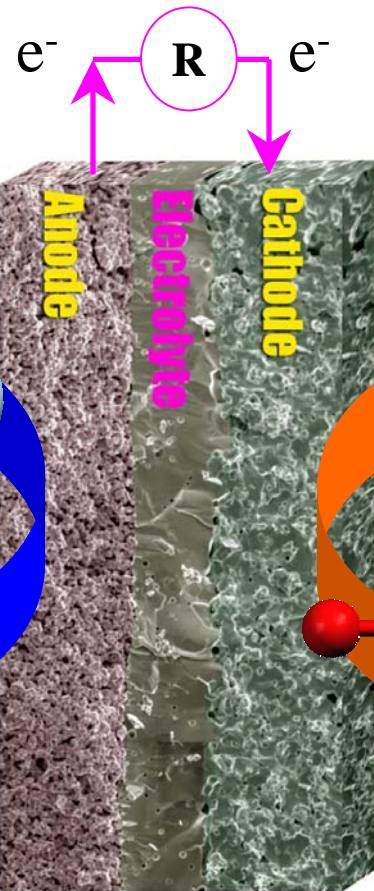
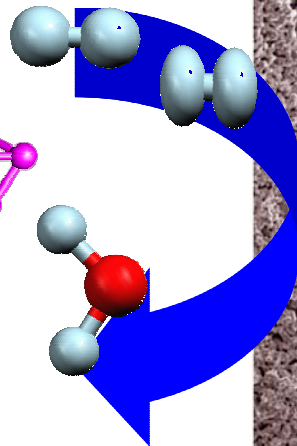
- The H<sub>2</sub>S Poisoning Effect
- Thermodynamic Analysis
- **QM Calculations**
- Exploration of New Sulfur-Tolerant Anode Materials

# Schematic of an SOFC in the molecular level

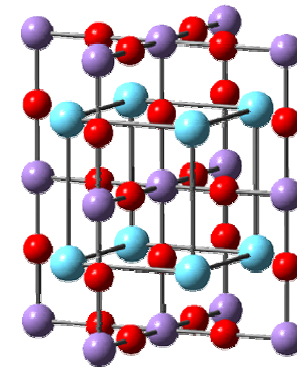
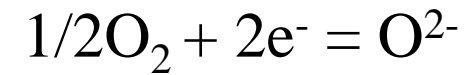
## Fuel oxidation



Ni-YSZ

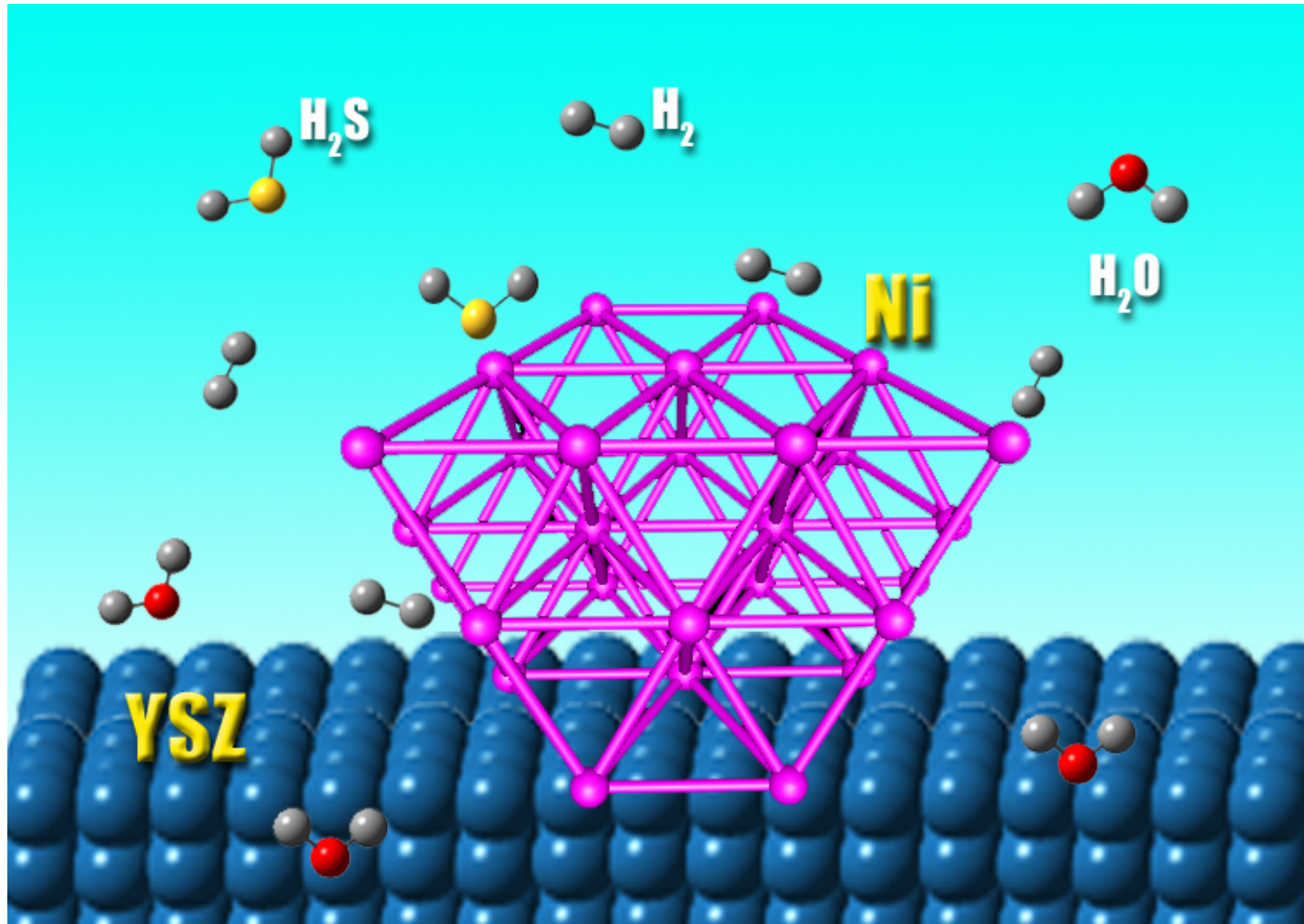


## Oxygen reduction



La<sub>1-x</sub>Sr<sub>x</sub>MnO<sub>3</sub>

# QM Calculation of H<sub>2</sub>S–Ni Interactions

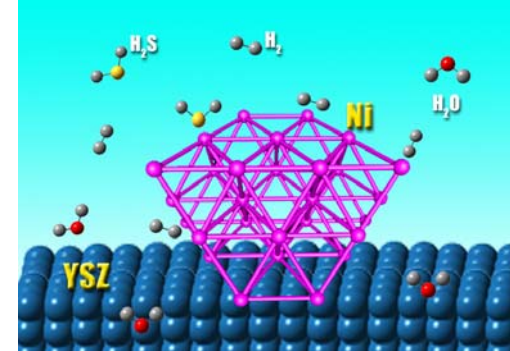


Sulfur-Tolerant Anodes

# Computational Approach

- Molecular properties of the gas molecules
- Properties of (e.g. crystal structure) of the solids
- Defect structures: vacancy, Interstitials, impurities

$$\mathbf{H}\Psi = \mathbf{E}\Psi$$



To Predict the Most Energetically Favorable Surface Configuration for reactants, intermediates, and products

**Energetics**



Energy for ads., dissociation, favorable reaction pathways

**Vibrations**

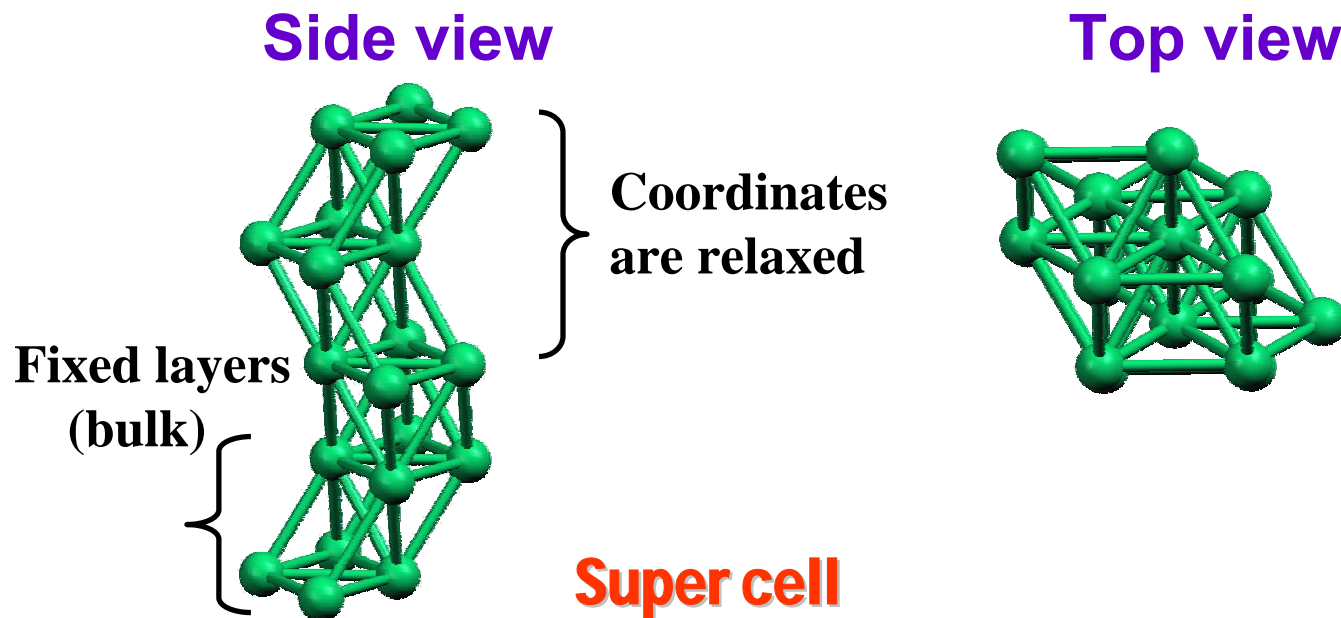


FTIR/Raman Spectroscopy

# Computational Method

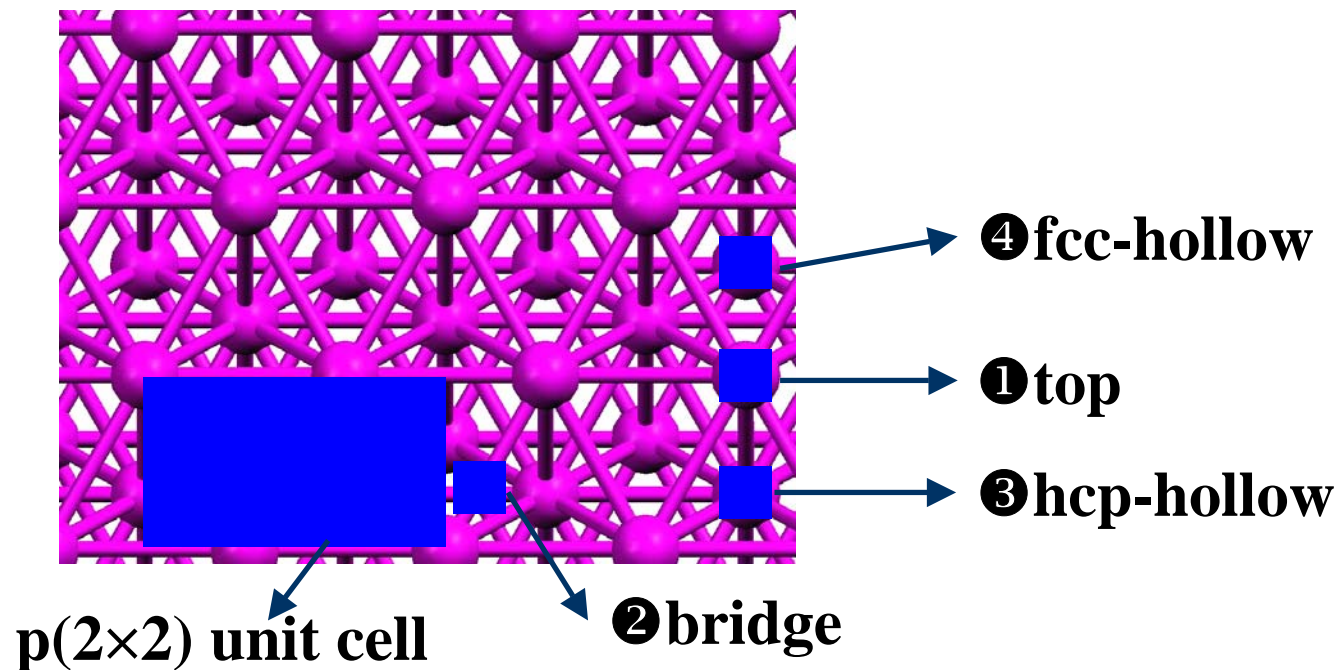
## ■ VASP (*Vienna Ab initio Simulation Package*)

- Supercell: Five layer [p(2×2)] units
- DFT: LDA with PW91 (GGA) correction
- Core pseudopotential
- Cut-off energy: 400 eV
- Vacuum space: ~ 10.0 Å





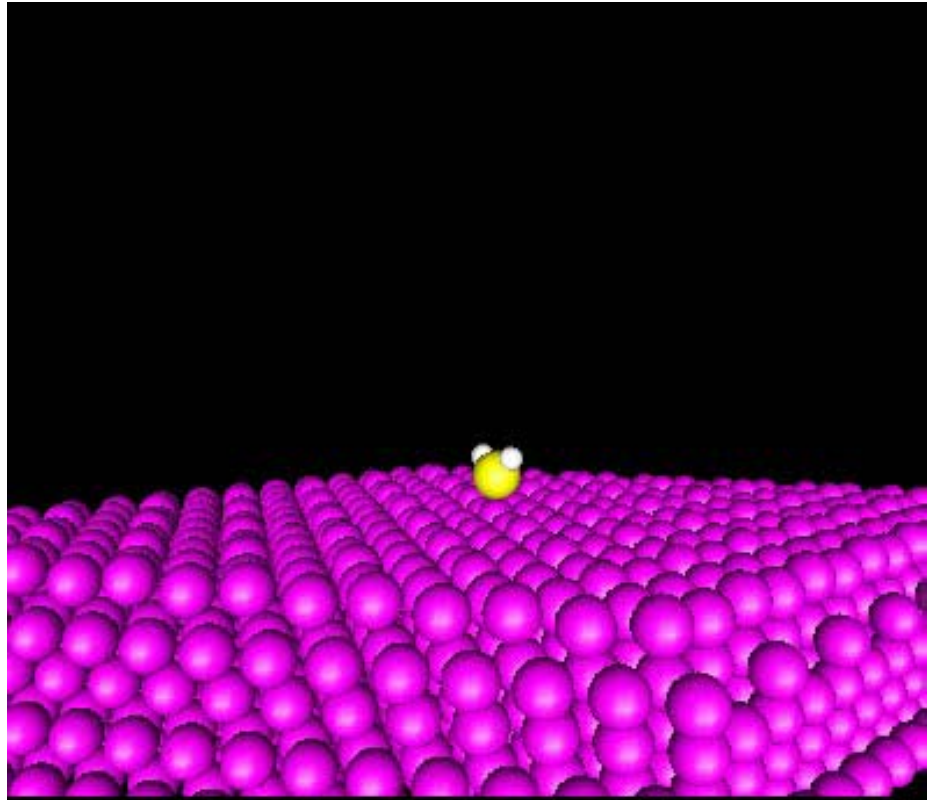
# (111) Surface and Adsorption Sites



## Adsorption energy in kcal/mol

$$\begin{aligned}\Delta E_{\text{ad}} &= \Sigma E[\text{products}] - \Sigma E[\text{reactants}] \\ &= E[\text{surface} + \text{adsorbate}] - E[\text{surface}] - E[\text{adsorbate}]\end{aligned}$$

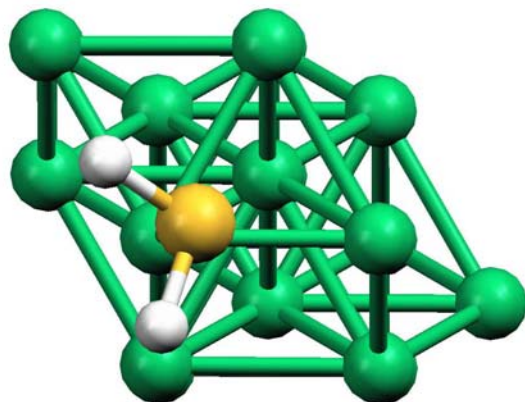
# Molecular Dynamics for H<sub>2</sub>S Dissociation



- H<sub>2</sub>S decomposition forming sulfur adsorption on Ni anode surface can occur in approximately **110 fs** under SOFC operating condition at 700°C.

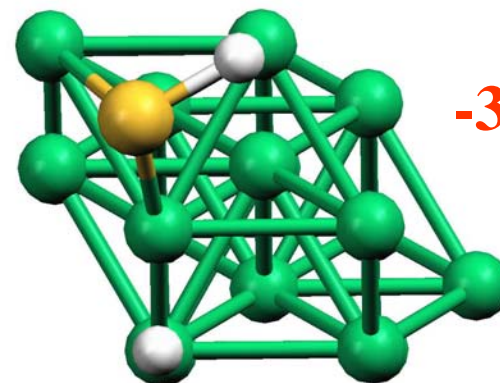
# H<sub>2</sub>S Decomposition on Ni(111) Surface

H<sub>2</sub>S Adsorption



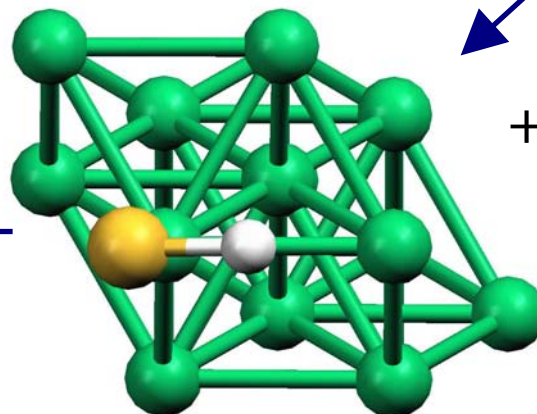
-13.1 kcal/mol

Dissociative H<sub>2</sub>S adsorption



-39.8 kcal/mol

SH + H



+ H

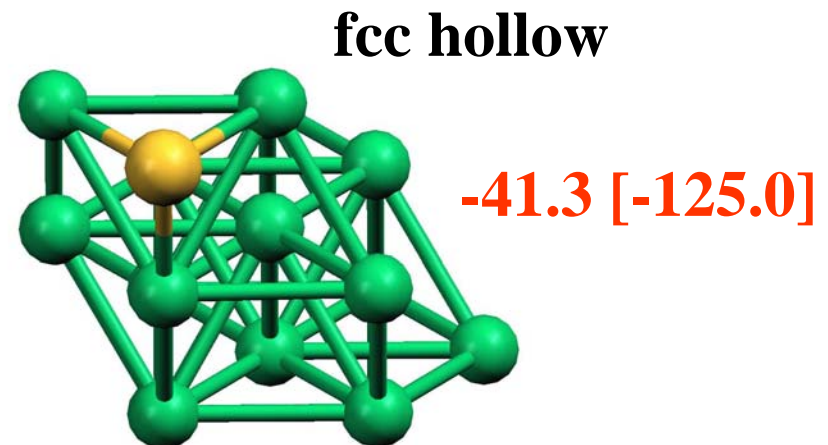
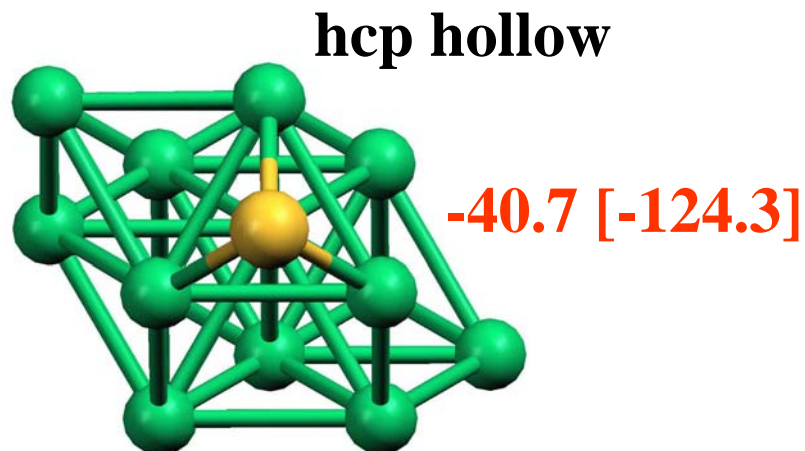
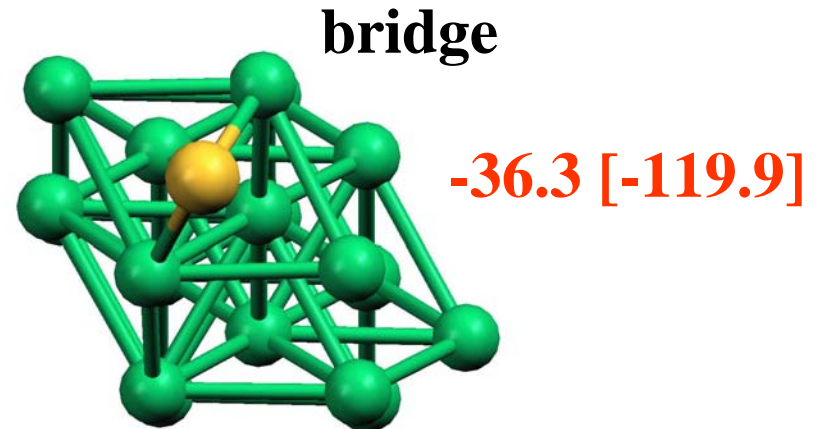
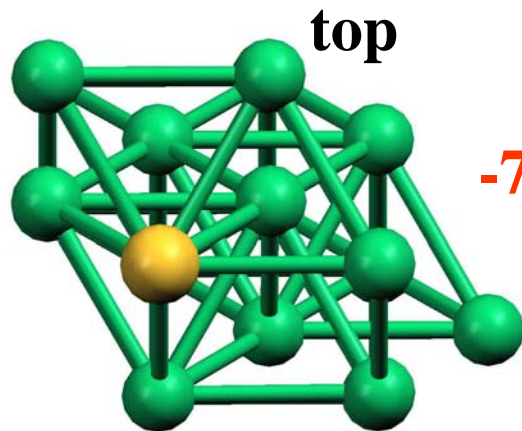
Ni(111)-S + 2H

-15.7 kcal/mol

- Energies are relative to Ni(111) + H<sub>2</sub>S



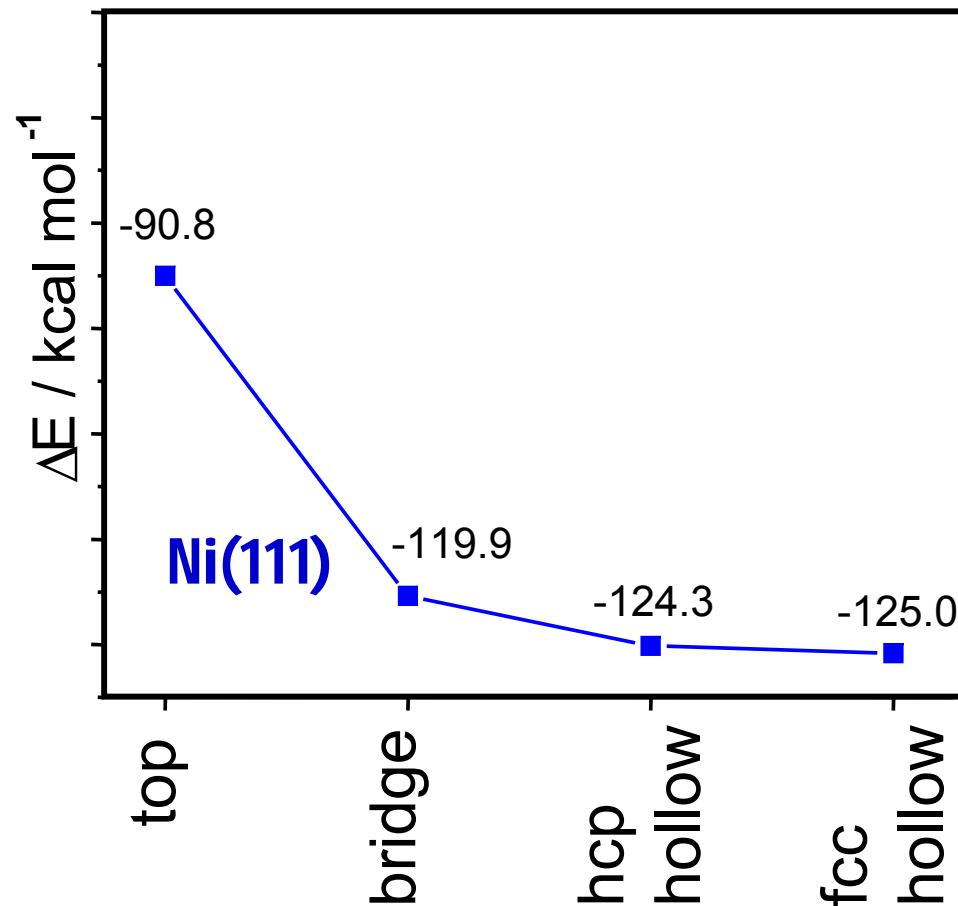
# Adsorption on Ni(111) Surface



- Energies in brackets are relative to Ni(111) + S

# Adsorption Energy on Ni (111) Surface

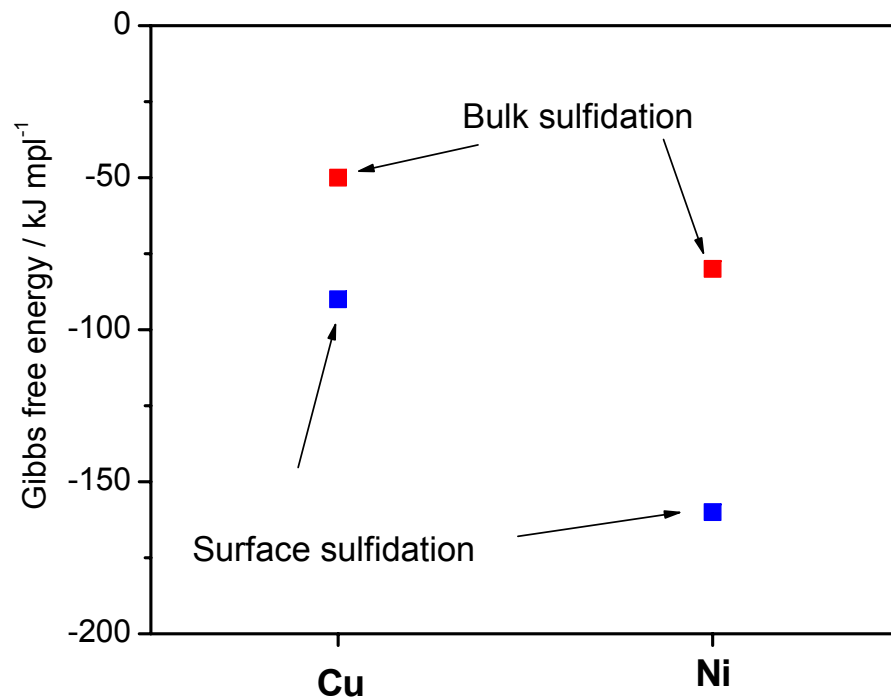
## ■ 1/4 monolayer (ML)



- Surface adsorption energy is very large
- Sulfur is very stable on Ni surface, difficult to remove

# Free Energies of Sulfide Formation on Ni & Cu

- ◆ Free energy change for sulfidation reaction at about 650°C

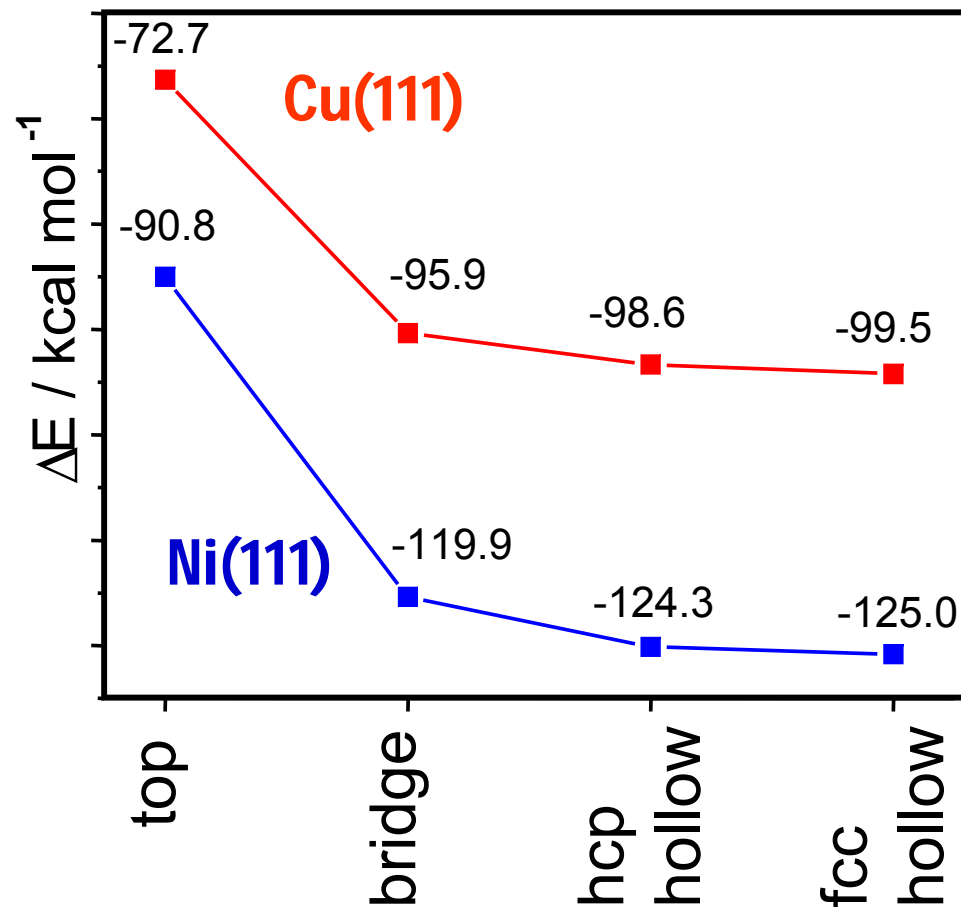


- The formation of bulk and surface sulfide on nickel is easier than on copper
- Surface sulfide is far more stable than bulk sulfide

- C. H. Bartholomew et al., *Advanced in Catalysis*, Vol. 31, pp. 166-170, **1982**.
- L. G. Marianowski et al., Eur. Patent, 88810131.8, **1988**.

# Sulfur Tolerance of Ni and Cu

## ■ 1/4 monolayer (ML)



- The results are in good agreement with data reported in literature
- Cu-based anodes is more sulfur-tolerant than Ni-based anodes

# Summary

- **Constructed Ni(111) surface for the slab model calculations**
- **Predicted adsorption energies of S on the Ni(111) and Cu(111) surfaces, suggesting that fcc hollow site is the most stable**
- **Predicted step-wise reaction mechanism of H<sub>2</sub>S decomposition on Ni(111) surface, which is in-line with XRD and Raman studies**

# Implications

- **While bulk sulfides may not be formed, sulfur strongly adsorbed on Ni or Cu surface blocks active sites for fuel oxidation, leading to performance degradation (poisoning effect)**
- **Surface adsorbed sulfur is difficult to remove, implying that Ni or Cu surfaces will have difficulty to get around sulfur poisoning effect**
- **New materials must be developed to achieve sulfur tolerance**

# Recent Progress: Since Oct 2004

- The H<sub>2</sub>S Poisoning Effect
- Thermodynamic Analysis
- QM Calculations
- **Exploration of New Sulfur-Tolerant Anode Materials**

# Requirements for S-Tolerant Anode Materials

## Two primary requirements

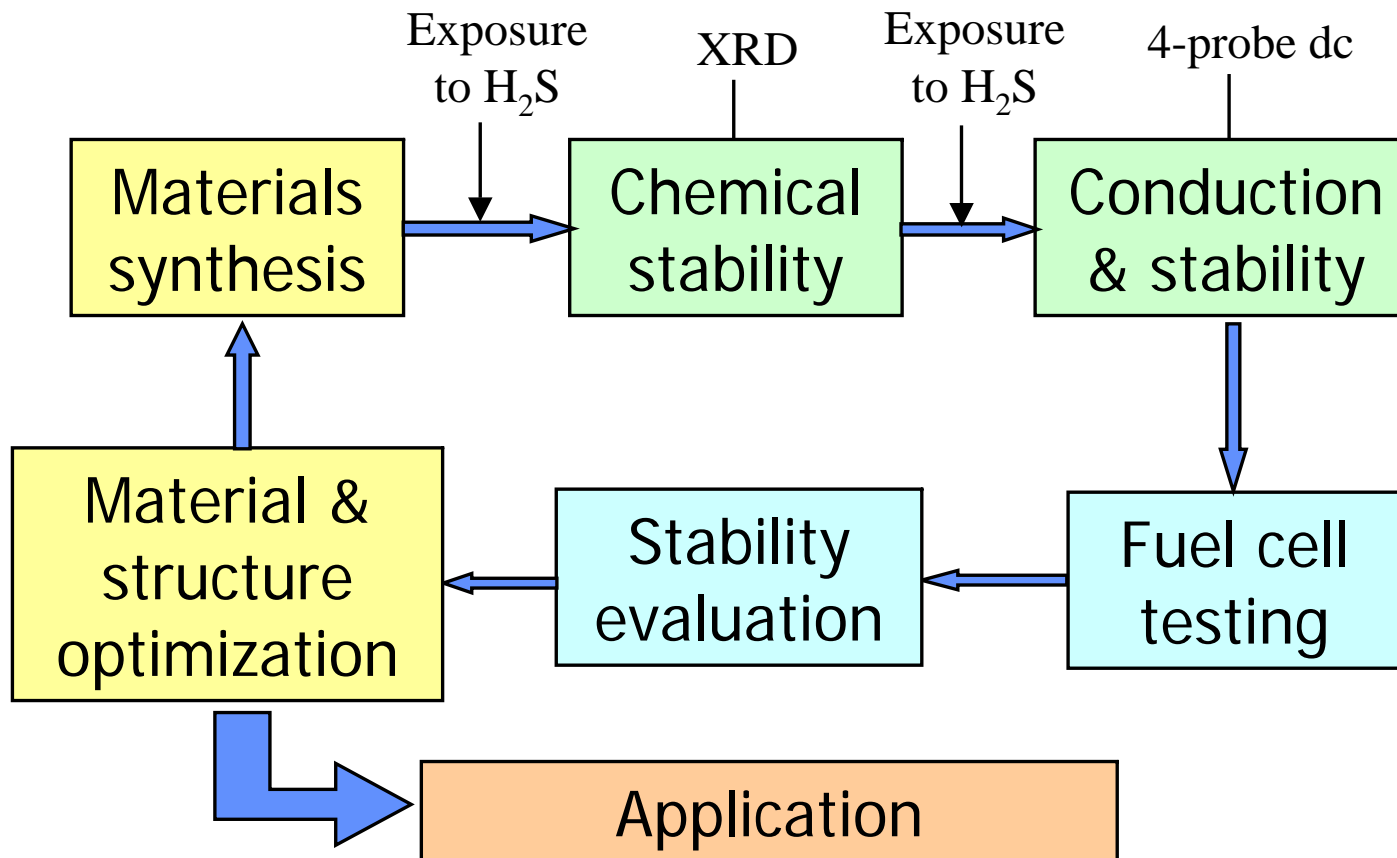
- Small Sulfur Adsorption Energy
  - High Catalytic Activity for Fuel Oxidation (H, C, S,...)
- \* For modification/decoration of Ni or Cu surfaces

## Other Desirable Properties

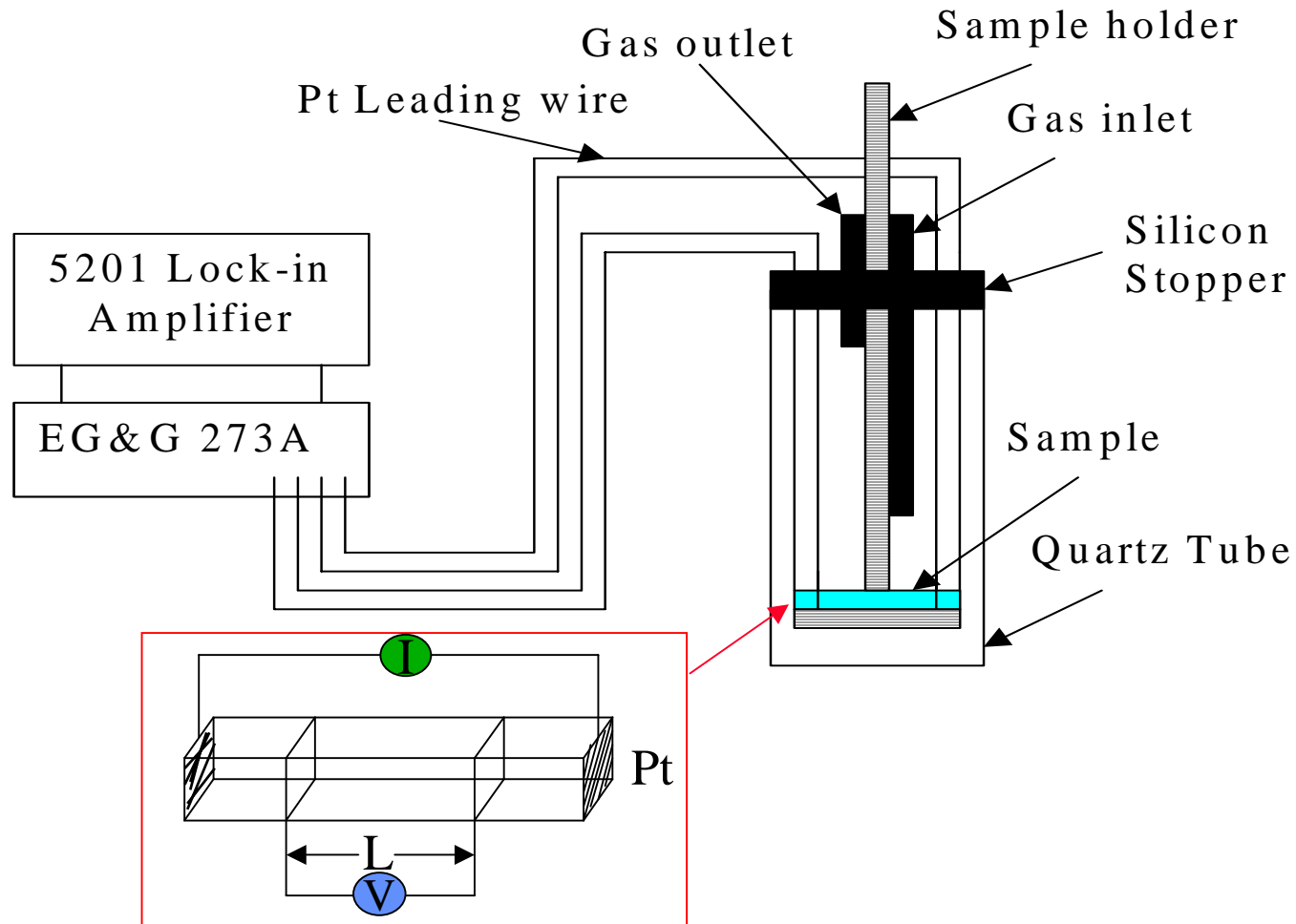
- Sufficient Electrical conductivity
  - Adequate compatibility with Electrolyte/Interconnect
  - Resistance to Oxidation
- \* For Replacement of Ni-Based anode



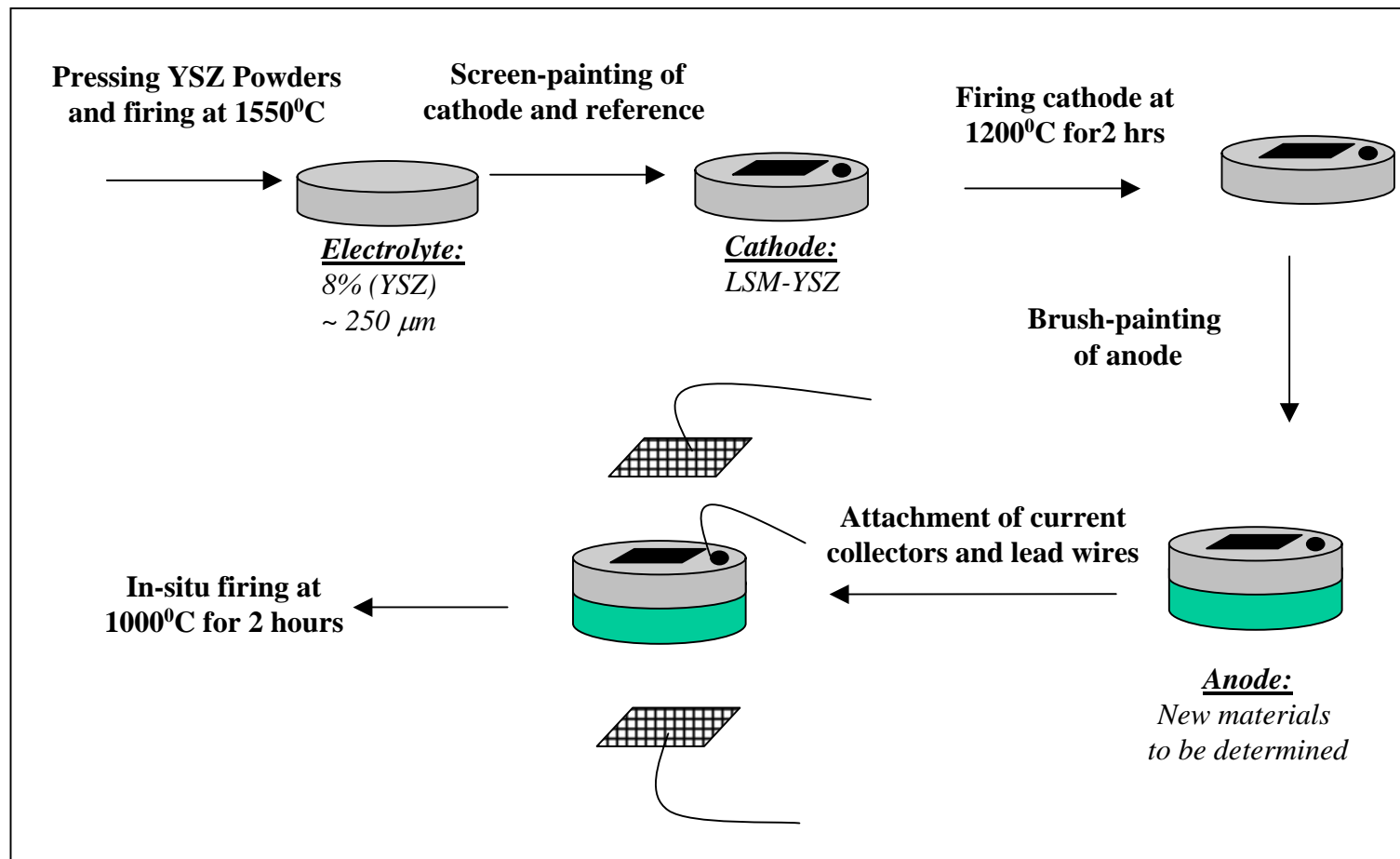
# Development of New Anode Materials



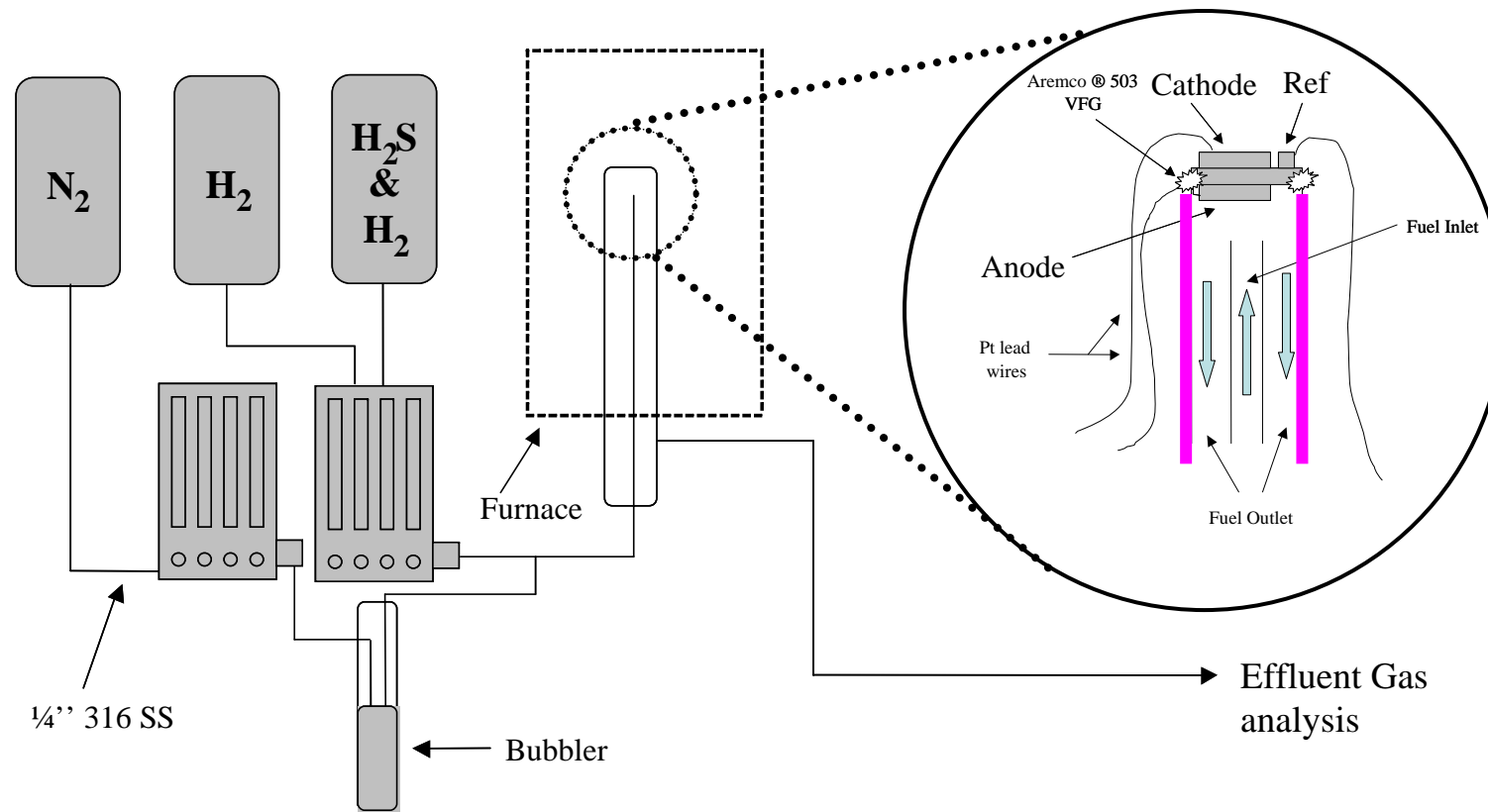
# Conductivity Measurement



# Fuel Cell Fabrication Procedures



# Testing of SOFCs in a H<sub>2</sub>S Containing Fuel



Experiment setup for testing of SOFC in H<sub>2</sub>S containing fuel

# Candidate Anode Materials

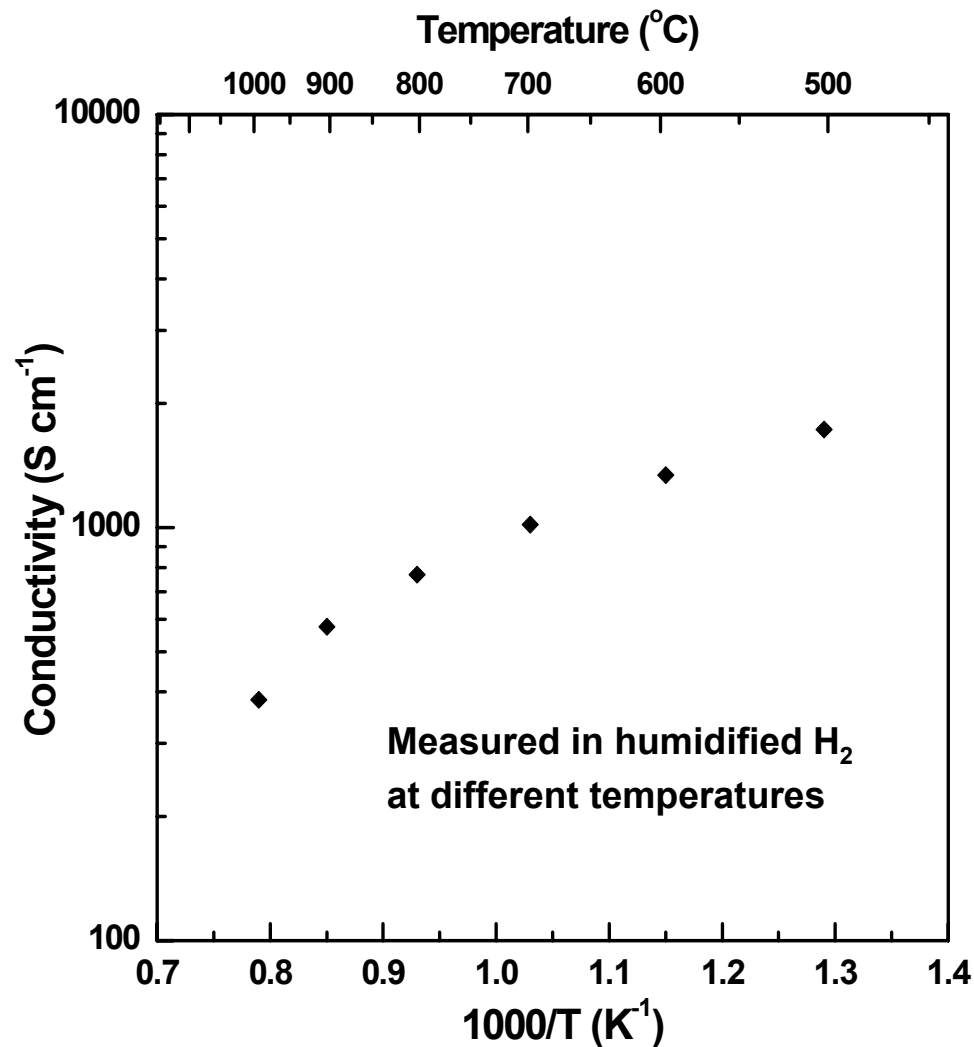
## Complex oxides (e.g., $ABO_3$ , $A_2B_2O_7$ , etc.)

- **Sufficient conductivity** due to delocalized  $d$  electron and/or doping-induced electronic defects
- **Good thermal match** with YSZ due to their relatively open structure
- **Good chemical stability** due to stabilized cations by the complex structure
- **Decent catalytic activity** due to the transition metal ions in the structure

Material for example	$\sigma$ at 800°C (S cm <sup>-1</sup> )	CTE (10 <sup>-6</sup> K <sup>-1</sup> )	$R_p$ at 800 °C ( $\Omega$ cm <sup>2</sup> )	Chemical stability
$La_{0.35}Sr_{0.65}TiO_3$ (LST) <sup>a, b, c</sup>	600	11-12	9	Excellent
$La_{0.7}Sr_{0.3}VO_3$ (LSV) <sup>a</sup>	140	10-12	4	Fair
$La_{1-x}Sr_xCr_{1-y}Mn_yO_3$ (LSCM) <sup>a, b, c</sup>	1.5	10-11	-	Excellent
$Gd_2Ti_{1-x}Mo_xO_7$ (GTMO) <sup>a</sup>	1	10.8	3	Fair
$SrVO_3$ <sup>a</sup>	800	-	1	Fair

<sup>a</sup> for GTFC, <sup>b</sup> for PNNL, <sup>c</sup> for Los Alamos

# Vanadium-Based Perovskites: $\text{SrVO}_3$



- **Vanadium-based oxide** has good catalytic activity towards sulfur, e.g.,  $\text{V}_2\text{O}_5$  is used for  $\text{SO}_2$  oxidation.
- **$\text{SrVO}_3$**  has high electrical conductivity:  $\sigma = 1000\text{S/cm}$  @  $700^\circ\text{C}$ , comparable to that for Ni/YSZ cermet

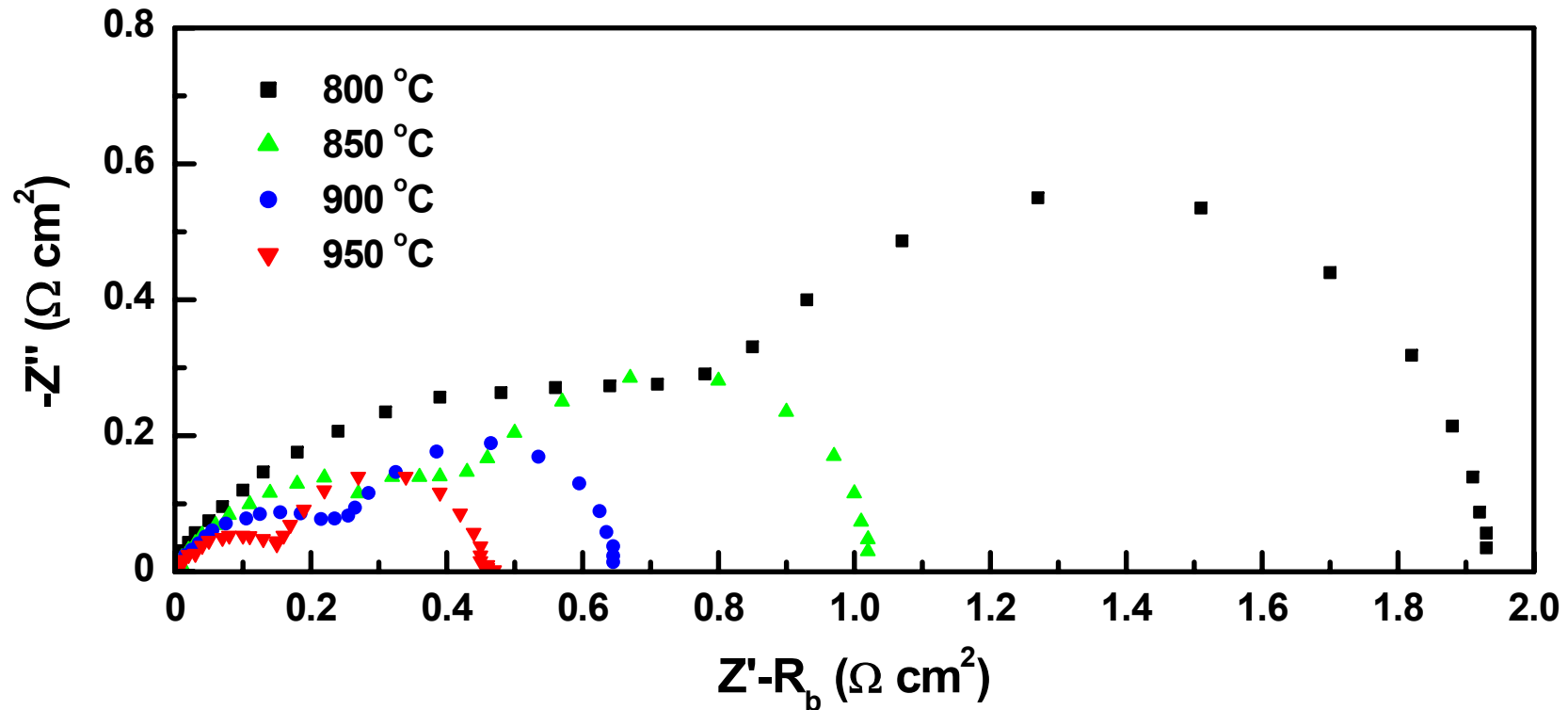
# Resistance to H<sub>2</sub>S

- Thermodynamic analysis indicated that **SrVO<sub>3</sub>** is chemically stable in 100 ppm H<sub>2</sub>S at elevated temperature.
- Under condition of 1000K, 100 ppm H<sub>2</sub>S/3%H<sub>2</sub>O/97%H<sub>2</sub>, the free energy change for the following sulfidation reaction:



is:  $\Delta G \approx + 37 \text{ kJ/mol}$ , i.e., this material is **thermodynamically stable against low concentration of H<sub>2</sub>S**.

# Polarization Resistance - SrVO<sub>3</sub>

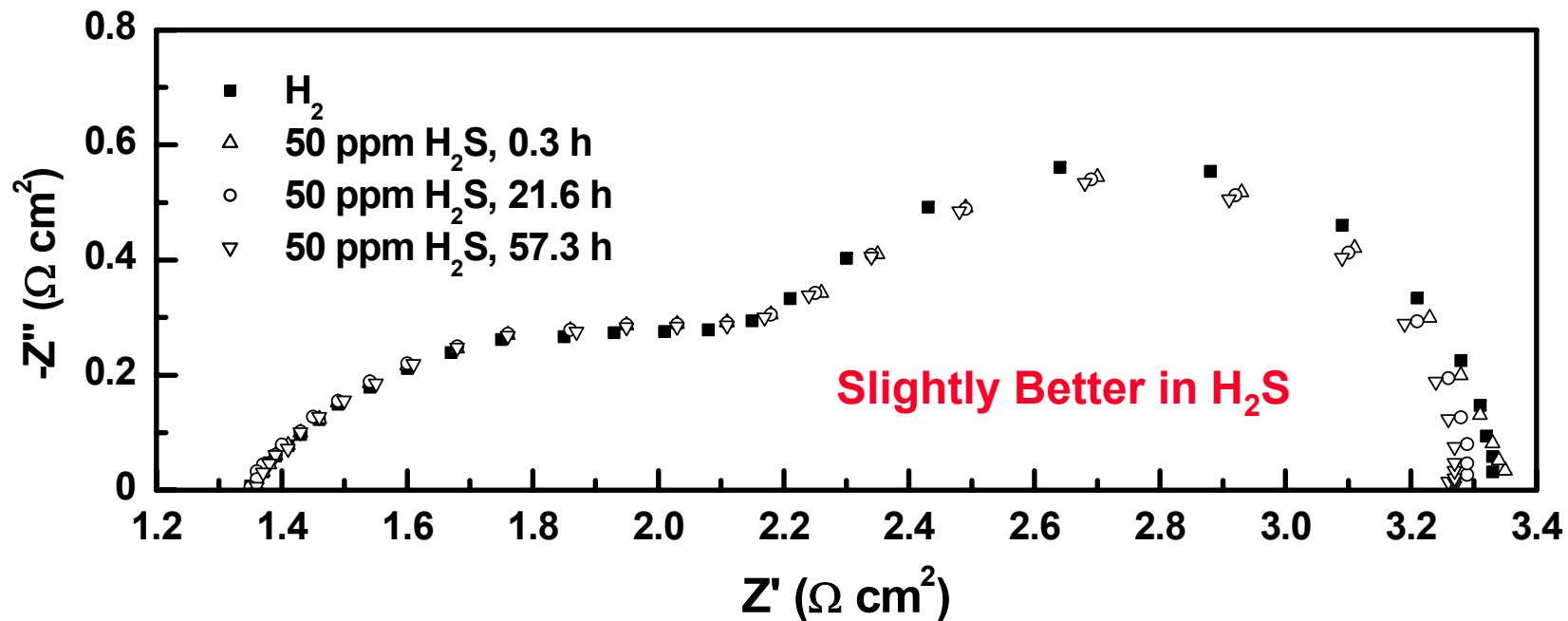


Impedance spectra for a SrVO<sub>3</sub>/YSZ/ SrVO<sub>3</sub> symmetrical cell in 50%H<sub>2</sub>/1.5%H<sub>2</sub>O/48.5%N<sub>2</sub>.

- SrVO<sub>3</sub> has decent activity for H<sub>2</sub> oxidation in fuel environment (0.25 Ω cm<sup>2</sup> @ 950°C); similar to Ni-YSZ



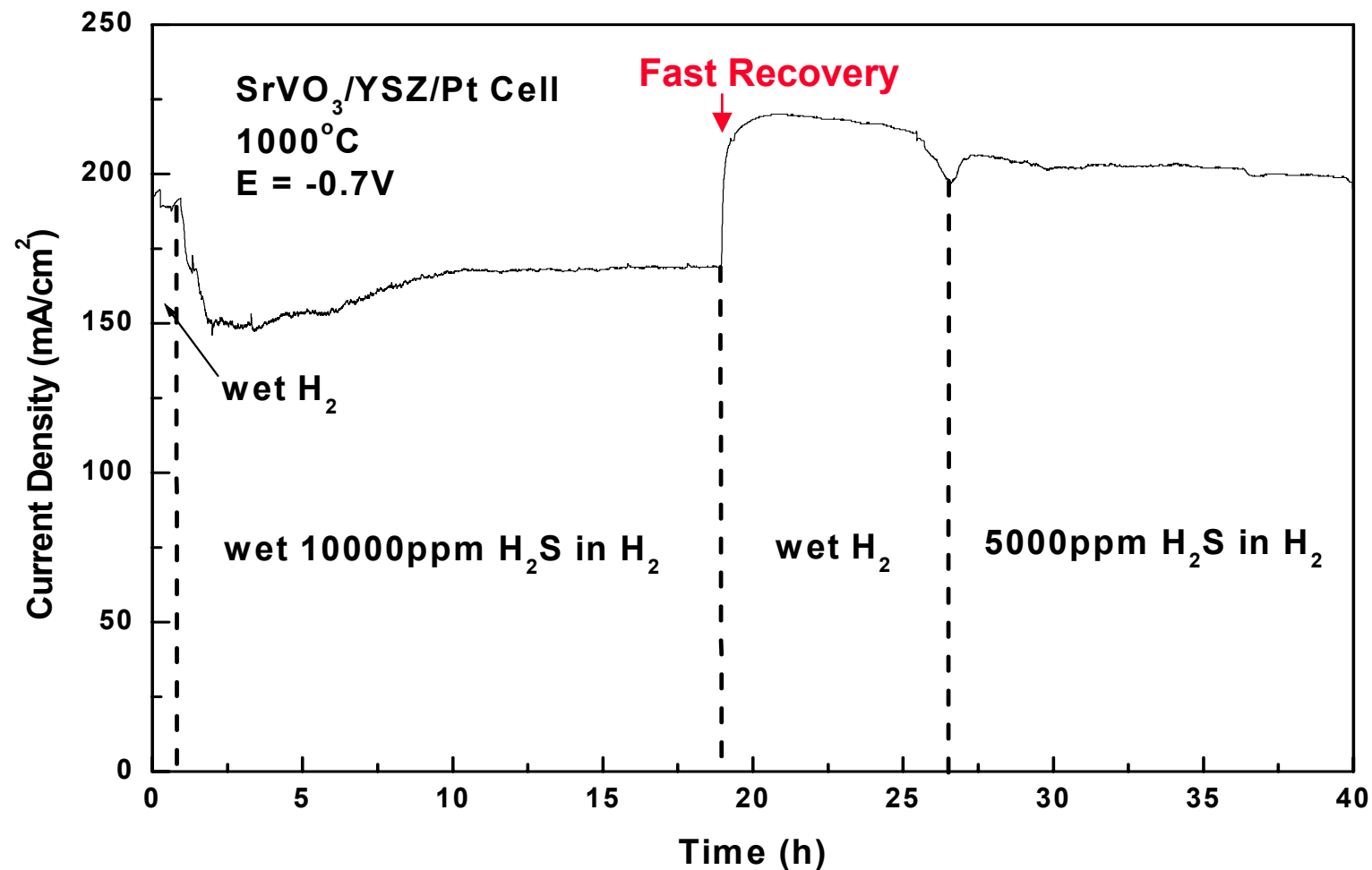
# Effect of 50 ppmv H<sub>2</sub>S on SrVO<sub>3</sub>



Impedance spectra for a SrVO<sub>3</sub>/YSZ/ SrVO<sub>3</sub> symmetrical cell when 50ppm H<sub>2</sub>S is introduced to a fuel of 50%H<sub>2</sub>/1.5%H<sub>2</sub>O/48.5%N<sub>2</sub>

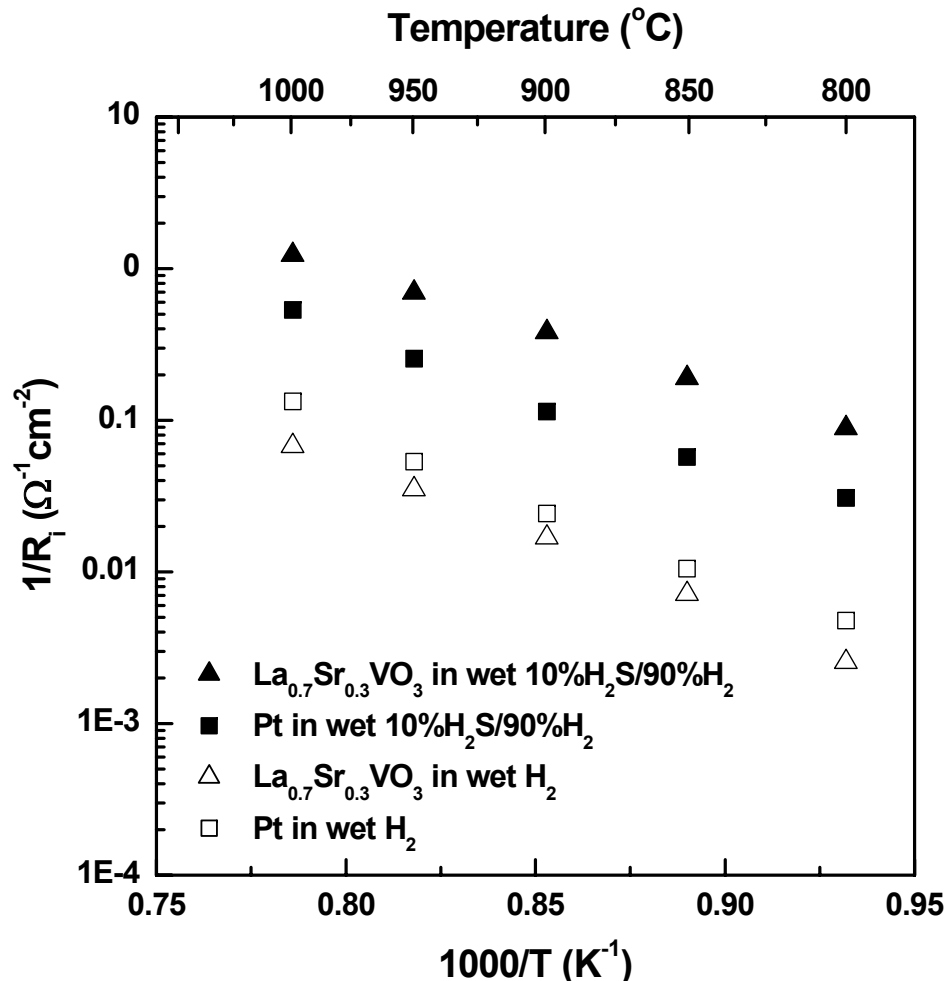
- The **interfacial resistance** for SrVO<sub>3</sub> anode showed **no degradation** in 50ppm H<sub>2</sub>S for ~60 h.

# Stability of SrVO<sub>3</sub>-Based SOFC in H<sub>2</sub>S/H<sub>2</sub>



The result of cell stability test under much harsher condition (i.e., 1000°C, 5000-10,000ppm H<sub>2</sub>S) suggested that the **performance loss** for SrVO<sub>3</sub> anode **might be reversible** up to **10,000 ppm** of H<sub>2</sub>S.

# Other Vanadium-Based Oxides: LSV



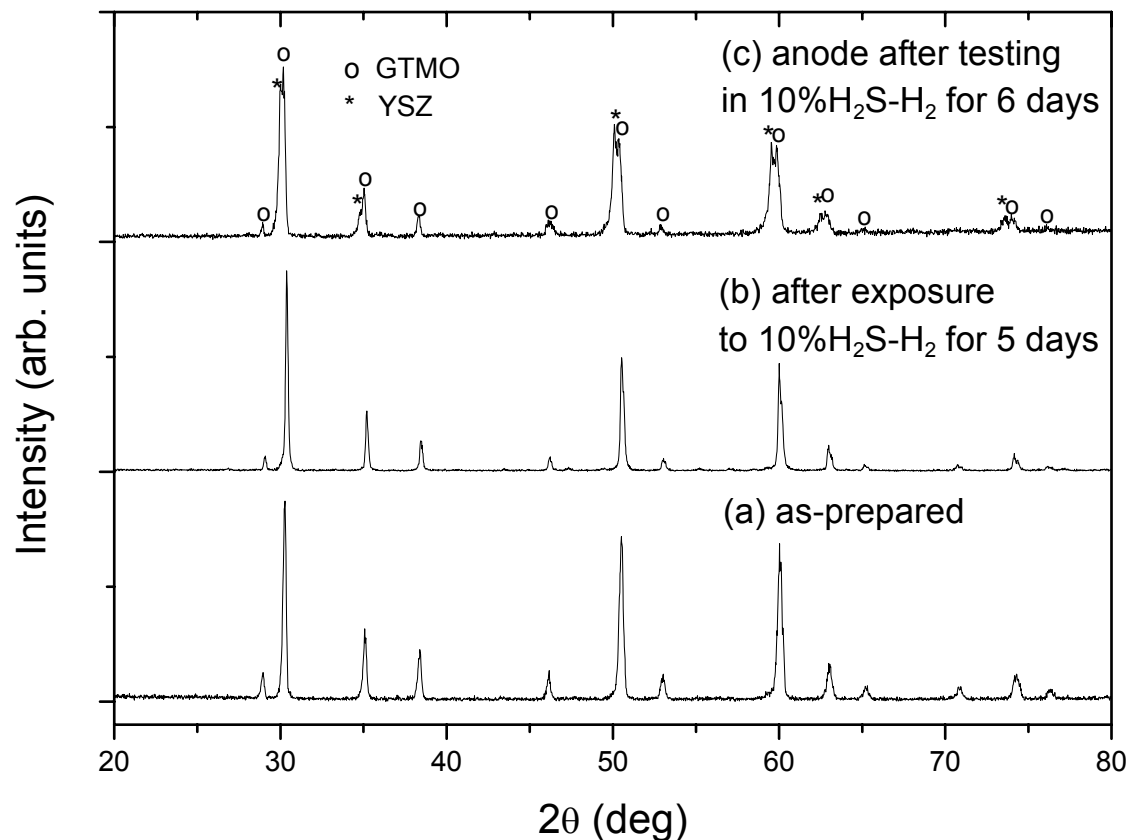
Inverse interfacial resistance vs. inverse temperature for LSV/YSZ/LSV and Pt/YSZ/Pt symmetrical cells at different temperatures

Sulfur-Tolerant Anodes

- **La<sub>0.7</sub>Sr<sub>0.3</sub>VO<sub>3</sub>** show even better activity in **H<sub>2</sub>S** than in pure H<sub>2</sub>
- The activity of LSV is **better than Pt** in high H<sub>2</sub>S content
- Possible candidate for **modifying the surface of Ni/YSZ anode**

# $\text{Gd}_2\text{Ti}_{1.4}\text{Mo}_{0.6}\text{O}_7$ (GTMO)

## Phase Stability in $\text{H}_2\text{S}$

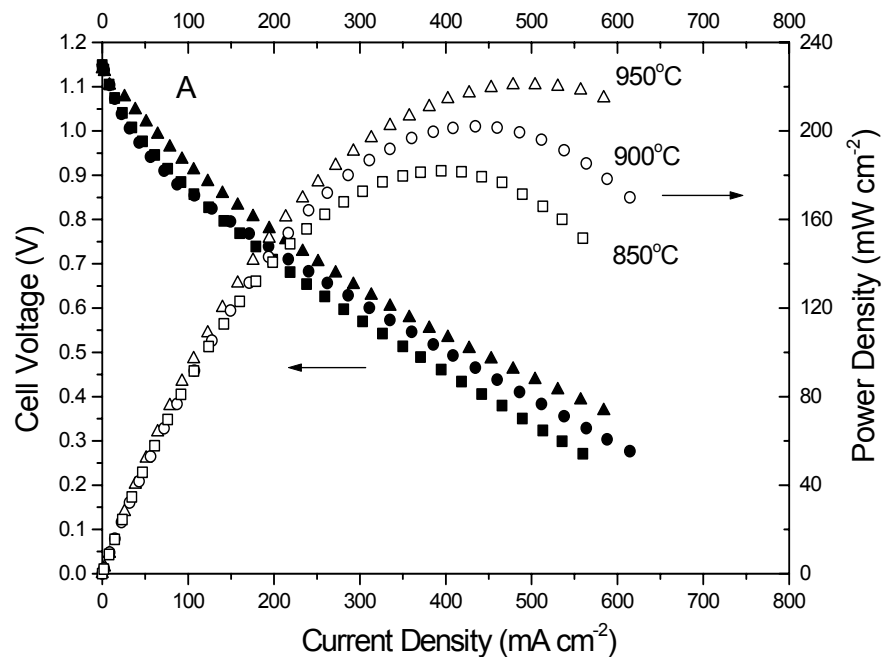


- Pyrochlore
- Phase stable under anodic conditions
- No reaction with  $\text{H}_2\text{S}$

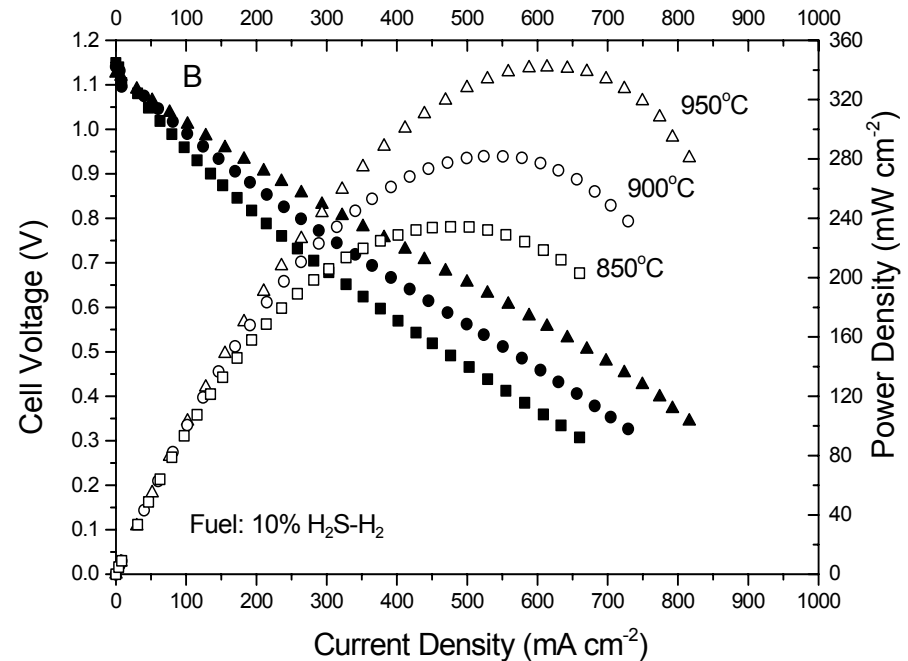
Note: Exposure & FC testing temperature: 950°C

# Performance of GTMO Anode in an SOFC

Fuel cell:  $\text{Gd}_2\text{Ti}_{1.4}\text{Mo}_{0.6}\text{O}_7/\text{YSZ}$  (0.25 mm)/LSCM



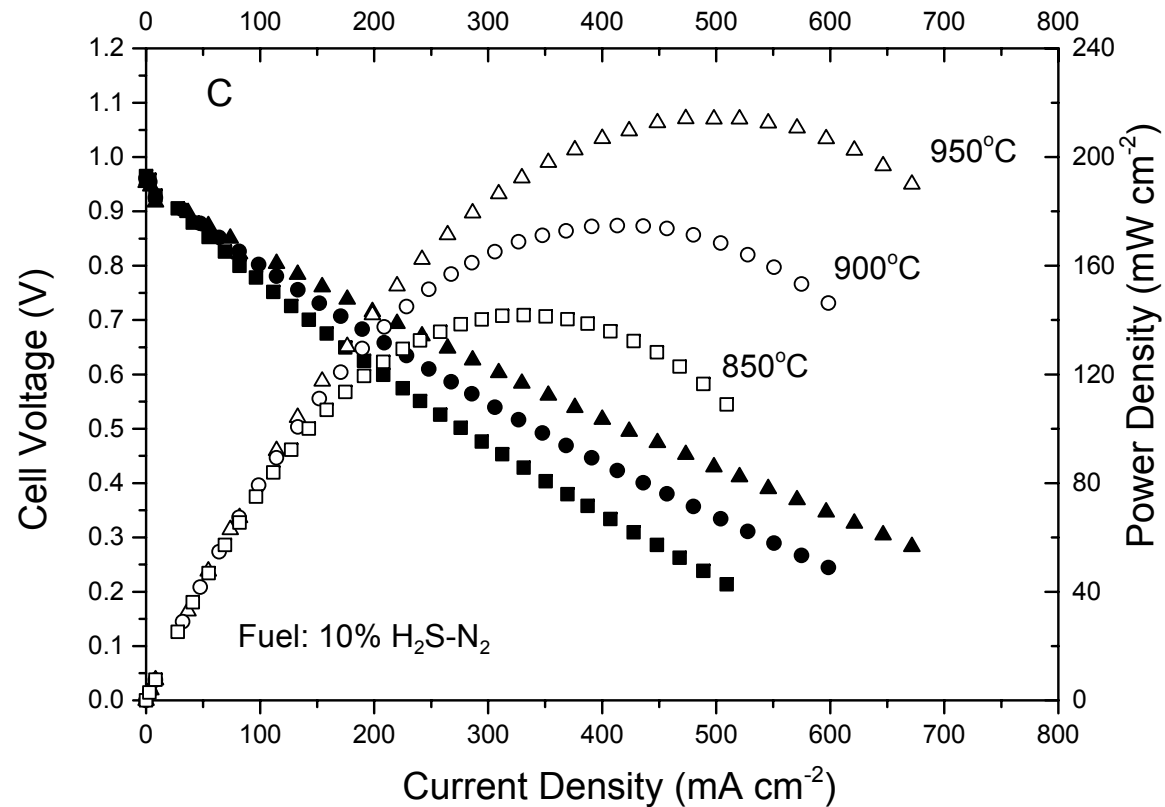
Fuel:  $\text{H}_2$ , 28 ml min<sup>-1</sup>



Fuel: 10%  $\text{H}_2\text{S}$ -90%  $\text{H}_2$ , 20 ml min<sup>-1</sup>

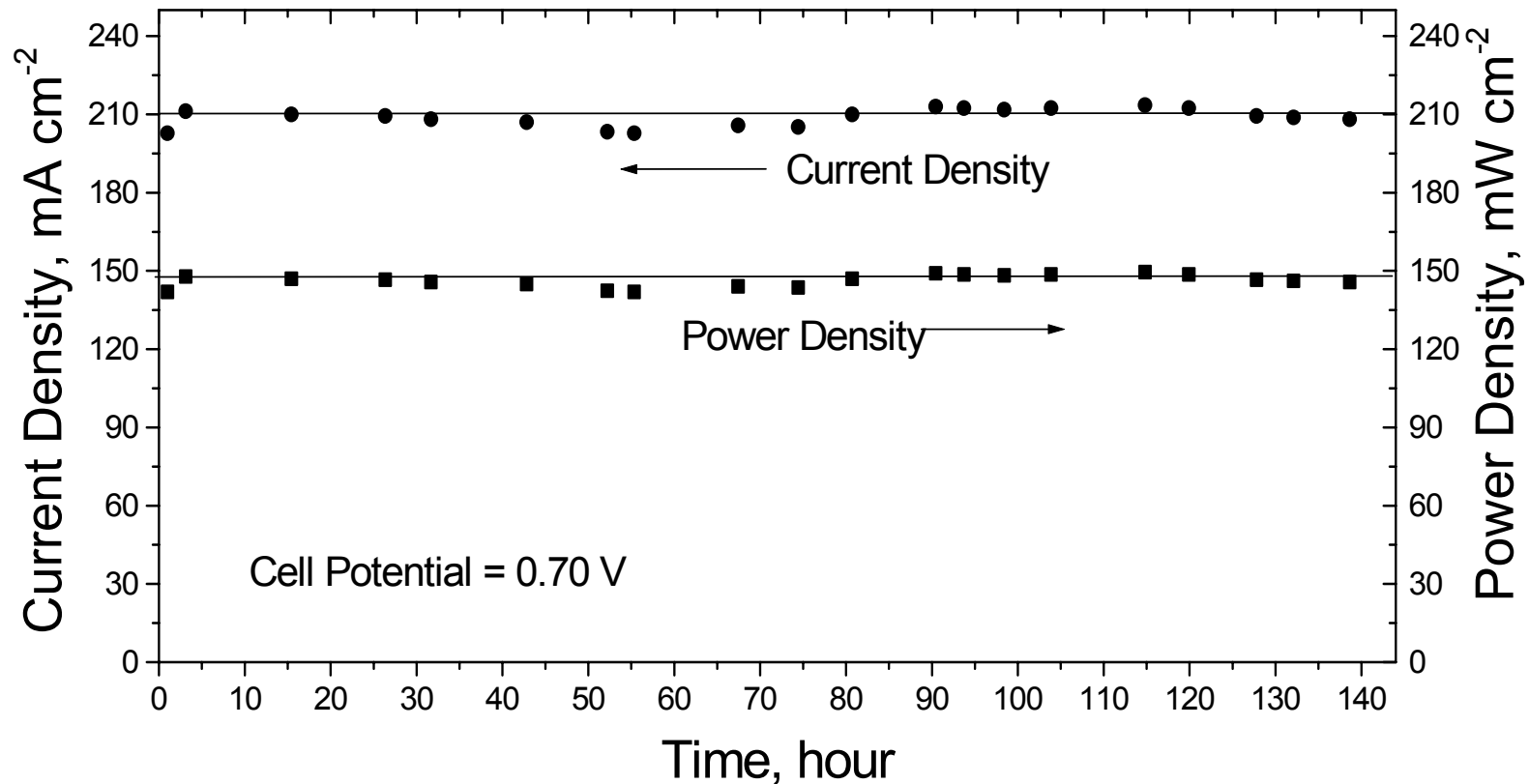
# Performance of GTMO in $H_2S$

Fuel cell:  $Gd_2Ti_{1.4}Mo_{0.6}O_7/YSZ$  (0.25 mm)/LSCM



Fuel: 10%  $H_2S$  in  $N_2$ , 12  $ml\ min^{-1}$

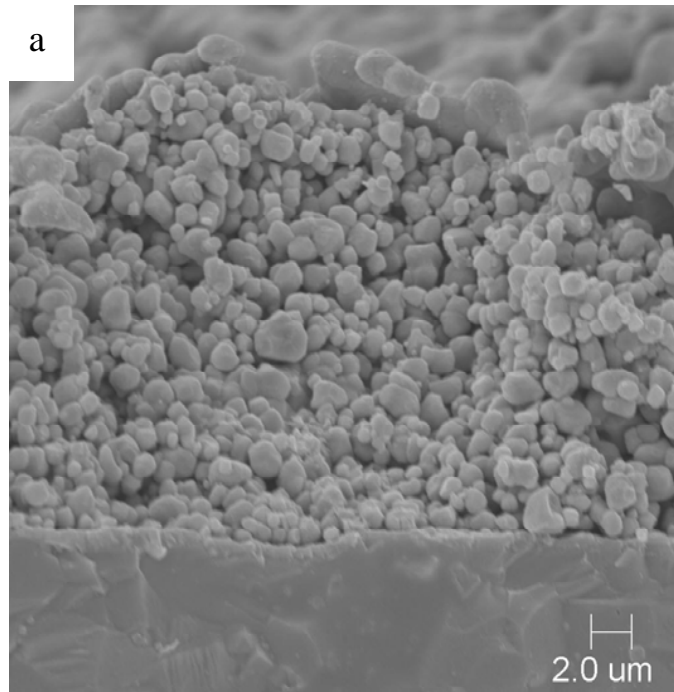
# Performance Stability of GTMO Anode



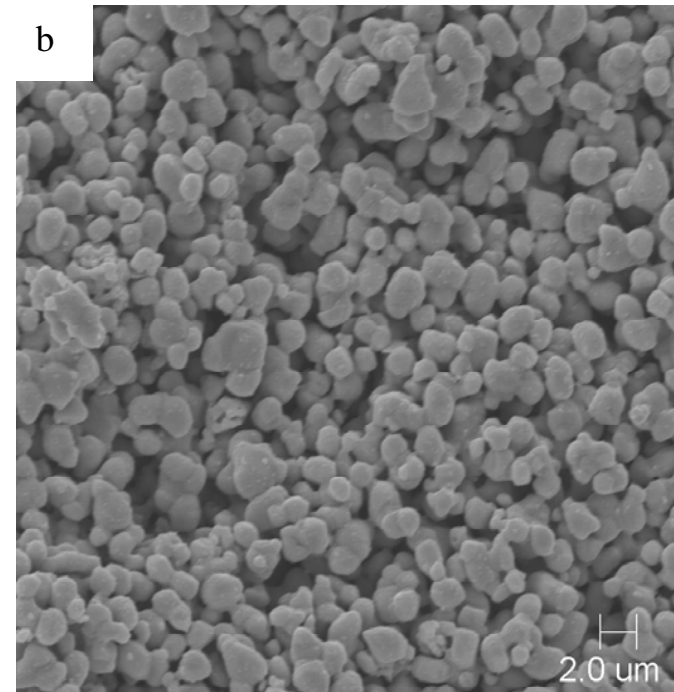
Stability testing (6 days) of a fuel cell operated on 10% H<sub>2</sub>S-H<sub>2</sub> at 950 °C at a constant cell terminal voltage of 0.70 V. Cathode: Pt

# Microstructure of $\text{Gd}_2\text{Ti}_{1.4}\text{Mo}_{0.6}\text{O}_7$ Anode

## Cross-Section



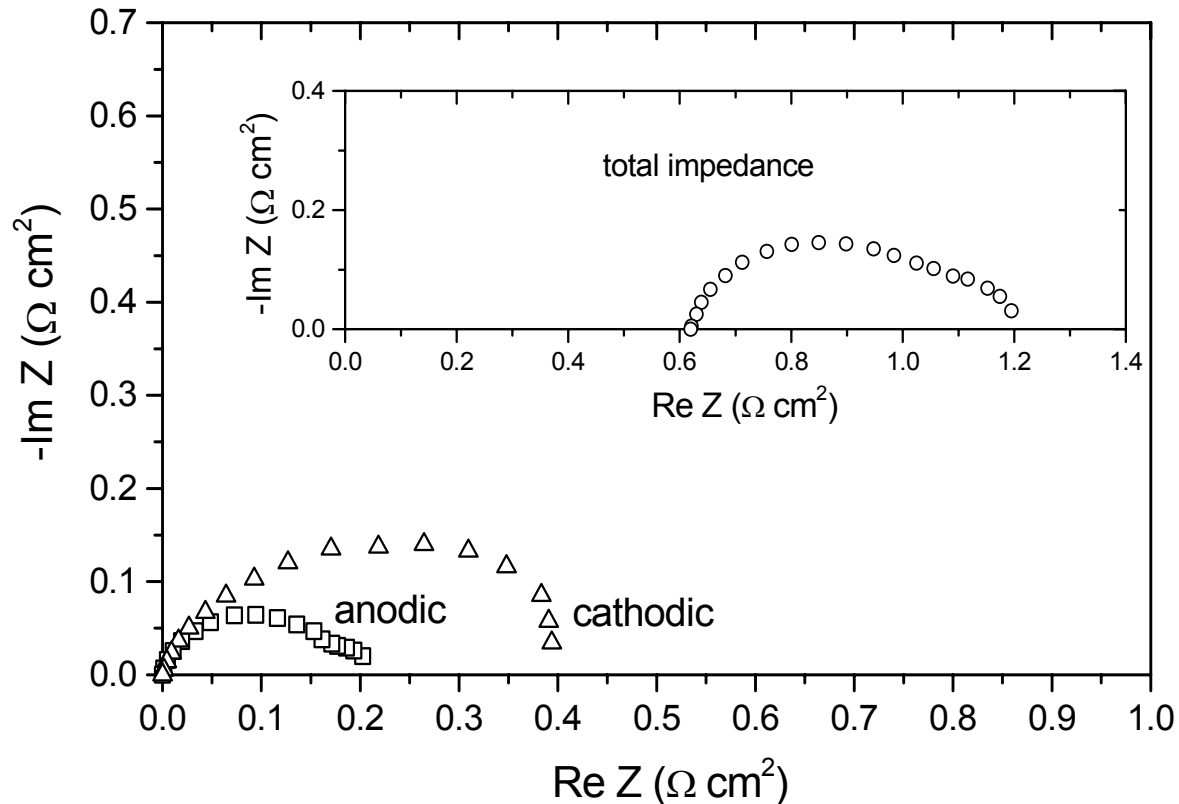
## Top View



SEM images of the  $\text{Gd}_2\text{Ti}_{1.4}\text{Mo}_{0.6}\text{O}_7$  anode after fuel cell testing



# Impedance Spectra of GTMO



YSZ bulk R:

0.6  $\Omega \text{ cm}^2$

Cathode/YSZ:

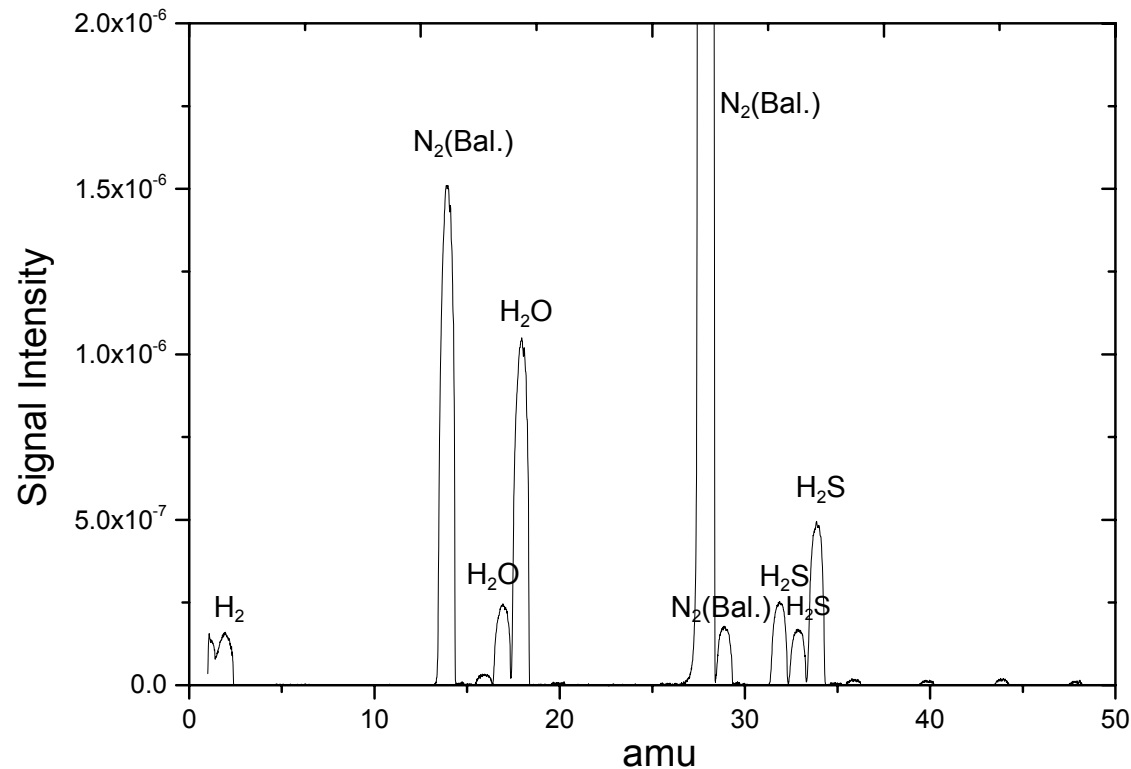
0.4  $\Omega \text{ cm}^2$

Anode/YSZ

0.2  $\Omega \text{ cm}^2$

Impedance spectra measured at 950 °C under open circuit conditions using a three-electrode configuration. Inset is the total impedance of the fuel cell. Anode: 10% $\text{H}_2\text{S}$ -90% $\text{H}_2$ , Cathode: air

# Effluent Gas Analysis



Effluent gas compositions:

$\text{H}_2\text{O}$ , small amount  $\text{H}_2$ , unreacted  $\text{H}_2\text{S}$ ,

and

element S (condensed in the gas line)

Mass spectrum of the effluent gas when the anode was fed on 10%  $\text{H}_2\text{S}$  ( $\text{N}_2$  balanced) at  $950^\circ\text{C}$ . The fuel cell was operated at a constant current density of  $400 \text{ mA cm}^{-2}$ . The fuel flow rate:  $12 \text{ ml min}^{-1}$ .

# Conclusions

- Vanadium based compounds demonstrated excellent S-tolerance and high catalytic activity in H<sub>2</sub>S-containing fuels and thus are possible candidates for modifying the surface of Ni/YSZ anode.
- Pyrochlore Gd<sub>2</sub>Ti<sub>1.4</sub>Mo<sub>0.6</sub>O<sub>7</sub> (GTMO) also showed excellent sulfur tolerance and catalytic activity towards electrochemical oxidation of H<sub>2</sub>S.

# Activities for the Next 6-12 Months

- To elucidation the mechanism for *H<sub>2</sub>S adsorption, decomposition, and interaction* with **oxide materials** using in-situ characterization techniques in order to achieve **intelligent design of new anode materials** with sulfur tolerance
- Further exploration of other potential sulfur-tolerant materials
- Evaluation of catalytic activities towards oxidation of other sulfur compounds/contaminants
- Long-term stability evaluation and strategies
- Surface and structure modification of conventional Ni-based anode to tolerate 50 ppm H<sub>2</sub>S
  - ✓ sputtering
  - ✓ solution infiltration
  - ✓ suspension infiltration

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